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Single β -Ga₂O₃ microbelt solar-blind photodetector with high specific detectivity, high rejection ratio and fast speed

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Abstract

An ideal solar-blind photodetector should possess high responsivity, low dark current, high speed, high spectral selectivity, high stability and a facile fabrication method. In this work, a high-quality single β -Ga₂O₃ microbelt photodetector is fabricated through low-pressure chemical vapor deposition and *in situ* atmospheric pressure annealing techniques. Due to the high quality and low defect density, this device shows a peak responsivity of 9.47 A W⁻¹ at 256 nm, a dark current of 1 pA and an ultrashort fall time of 1.37 μ s under 20 V bias. Moreover, an ultrahigh specific detectivity of ~10¹⁴ Jones and an extremely large solar-blind/ultraviolet A rejection ratio of nearly 10⁵ have also been achieved, suggesting the excellent sensitivity and wavelength selectivity of our single β -Ga₂O₃ microbelt photodetector. Besides that, the β -Ga₂O₃ microbelt photodetector exhibits excellent mechanical and long-term stabilities. Our findings provide a facile and promising route to develop high performance solar-blind UV photodetectors.

Keywords: β-Ga₂O₃ microbelt, solar-blind photodetector, high quality, high performance

(Some figures may appear in colour only in the online journal)

1. Introduction

Since the solar radiation at 200–280 nm (solar-blind ultraviolet) is strongly absorbed by ozone in the atmosphere and cannot reach the earth's surface, the photodetectors working in this waveband, widely known as solar-blind photodetectors, have the advantages of low background interference and high sensitivity, which have been widely used in military and civil fields [1–7]. An ideal solar-blind UV detector should have high responsivity, high spectral selectivity, low dark current and high response speed [8]. Compared with conventional photomultiplier tubes and Si-based ultraviolet photodetectors, solar-blind ultraviolet photodetectors based on wide-band gap semiconductors (Ga₂O₃, AlGaN, ZnGa₂O₄, ZnMgO, diamond, etc) offer special superiorities such as all solid state, low

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power consumption and no filter required, which have become a current research hotspot [9-19]. Among these wide band-gap semiconductors, Ga_2O_3 -related materials, especially β -Ga₂O₃ have attracted the increasing attention due to their intrinsic solar-blind band gap (4.7–5.1 eV), high radiation hardness and excellent thermal and chemical stability [20, 21]. To date, various device geometries including metal-semiconductor-metal (MSM) photodetectors, heterojunction/Schottky barrier photodiodes, and phototransistors have been demonstrated on Ga₂O₃ bulk, thin films or nanostructures [22-26]. Surprisingly, high responsivity (>1 A W^{-1}) has been widely achieved in different types of Ga₂O₃-based photodetectors [27-29]. Nevertheless, present Ga₂O₃ solar-blind photodetectors commonly suffered from slow response speed, large dark current, and low solar-blind/UV rejection ratio, which may be related to high sub-band-gap response, strong trapping effect, and high donor concentration caused by low crystalline quality and extensive oxygen vacancies of Ga2O3. Although homoepitaxy and high temperature oxygen annealing have been commonly used to improve the crystal quality and reduce the oxygen vacancy density, they still cannot meet the needs of high-speed, low dark current and high rejection ratio UV detection [30-32].

Compared to bulk and thin film counterparts, the materials that naturally grown into one-dimensional (1D) nano/ microstructures are more likely to achieve high crystalline perfection and reduced defects [33-35]. Moreover, 1D nano/microstructured semiconductors also have the advantages of large specific surface area, light weight, good flexibility, and low cost. Therefore, solar-blind UV detectors based on 1D β -Ga₂O₃ have recently emerged as one of the most promising approaches to improve the photodetection performance [29, 36–46]. In particular, β -Ga₂O₃ nano/microbelts (sheets or ribbons) photodetectors have been highlighted for their high responsivity ($\sim 10^2 - \sim 10^4 \text{ A W}^{-1}$) [29, 34, 37, 47, 48]. However, the reported solar-blind photodetectors based on 1D β -Ga₂O₃ nano/microbelts commonly suffer from the low solar-blind/UVA (ultraviolet A) rejection ratio ($\sim 10^2 - \sim 10^4$) and slow response speed (~ 0.1 few seconds), which should be associated with surface defects and polycrystalline formation [41, 45, 49]. As a result, the performance of solarblind UV detectors based on 1D β -Ga₂O₃ is still lower than expected.

In this paper, high-quality single-crystal β -Ga₂O₃ microbelts are grown by low-pressure chemical vapor deposition (CVD) and *in-situ* atmospheric pressure annealing technologies. The β -Ga₂O₃ microbelts show high crystalline quality, few oxygen vacancies and very flat surface. The MSM structure solar-blind UV detector is fabricated on an individual β -Ga₂O₃ microbelt with 100 μ m channel length. The fabricated device shows a low dark current of 10^{-12} A, a peak responsivity of 9.47 A W⁻¹ at 256 nm and a fast decay time of 1.37 μ s. Moreover, the device has almost no response to 365 nm illumination with a highest solar-blind/UVA rejection ratio ($R_{\text{peak}}/R_{365 \text{ nm}}$) of nearly 10⁵. In addition, the device could keep high performance under bending state.

2. Experimental details

One-dimensional β -Ga₂O₃ microbelts were grown via a lowpressure CVD and in-situ atmospheric pressure annealing technologies. To prepare Ga2O3 microbelts, Ga2O3 and graphite (C) high-purity powers (99.99%) with a weight ratio of 1:1 served as the precursor mixtures. Ga₂O₃/graphite mixed powder and oxygen gas were used as Ga and O sources, respectively. The mixed powder was placed in a corundum boat and then slowly pushed into a tube furnace. After that, the tube furnace was pumped down to a base vacuum using a mechanical pump and heated up to 900 °C at a speed of 15 °C min⁻¹. At this point (900 °C), the mechanical pump was turned off. Subsequently, Ar and O₂ gases with a flow ratio of 6:1 were introduced into the tube, so that the pressure in the tube reached ~ 0.05 MPa. Then the temperature was slowly increased at 3 °C min⁻¹, and at the same time, β -Ga₂O₃ microbelts began to grow. And the following reactions might take place in the tube furnace:

$$C(s) + Ga_2O_3(s) \rightarrow Ga(g) + CO_2(g)$$
(1)

$$Ga(g) + O_2(g) \rightarrow Ga_2O_3(s).$$
⁽²⁾

After 60 min, the temperature was raised to 1080 °C, and the growth of the microbelts was completed. Subsequently, Ar and O₂ gases were continuously introduced into the tube and the pressure inside the tube was increased and maintained at atmospheric pressure. Then the microbelts were annealed *in-situ* at 1080 °C in Ar and O₂ atmospheres for 60 min at the atmospheric pressure to improve the crystalline quality and repair oxygen vacancy defects. Finally, the tube furnace was slowly cooled down to room temperature, and the highquality β -Ga₂O₃ microbelts can be obtained at the edge of corundum boat (figure 1(a)). The MSM photodetector was fabricated on an individual β -Ga₂O₃ microbelt by sputtering 30 nm thick gold electrodes using a shadow mask on a flexible substrate, and the distance between two electrodes is 100 μ m.

The characterizations of the Ga₂O₃ microbelts were investigated by scanning electron microscopy (SEM, HITACHI S-4800), transmission electron microscopy (TEM, FEI Talos F200s), x-ray diffraction (XRD, Bruker D8GADDS) with Cu K α radiation ($\lambda = 0.154$ nm), x-ray photoelectron spectroscopy (XPS) spectrometer (Thermo ESCALAB 250) and Raman spectrometer (Finder Vista) with a 532 nm laser. Current–voltage (*I–V*) and time–dependent current (*I–t*) curves of the photodetector were obtained using a semiconductor device analyzer (Agilent B1500A). The photoresponse spectrum was measured through responsivity measurement system equipped with an SR 830 lock-in amplifier and a 150 W Xe lamp. The transient responses were recorded by a Tektronics DPO5104 oscilloscope under 260 nm laser excitation with a pulse width of 10 ns.



Figure 1. (a) Photograph, (b) optical micrograph, (c) SEM images, (d) high-resolution TEM (HRTEM) and selected area electron diffraction (SAED) images, (e) XRD pattern, (f) Raman spectroscopy and (g) O1s XPS spectrum of the β -Ga₂O₃ microbelt.

3. Results and discussion

Figures 1(a)–(c) show the photograph, the optical micrograph and the SEM image of the as-prepared materials, indicating that these products are available in microbelt shapes with typical widths of 5–120 μ m, thicknesses of 0.1–2 μ m and lengths of 1-12 mm. Clearly, the surface of the microbelts is very flat and smooth. Figure 1(d) shows a high-resolution TEM (HRTEM) image taken from the edge of the thin Ga₂O₃ microbelt. The distances between adjacent layers are about 0.30 and 0.24 nm, corresponding to (010) and (310) planes of β -Ga₂O₃ with a monoclinic structure, respectively. The corresponding selected area electron diffraction (SAED) image of Ga_2O_3 microbelt is shown in the inset of figure 1(d). Figure 1(e) presents the XRD pattern of multiple β -Ga₂O₃ microbelts laid flat on a sample stage. Two strong diffraction peaks at $2\theta = 30.1^{\circ}$ and 45.9° correspond to the (400) and (600) crystal faces of monoclinic β -Ga₂O₃, respectively (JCPDS card No. 43-1012), and the peak positions are in agreement with the reported values of the single-crystal bulk β -Ga₂O₃ [25, 50, 51]. The inset of figure 1(e) shows a magnified (400) diffraction peak of β -Ga₂O₃ microbelts and its full-width half maximum is estimated to be $\sim 0.113^{\circ}$. No extra peaks can be observed in the XRD spectrum. HRTEM, SAED and XRD results suggest that the length and thickness directions of these Ga_2O_3 microbelts are along the [010] and [100] orientations, respectively. To further analyze the structural property of β -Ga₂O₃ microbelts, Raman spectroscopy was performed on a single microbelt at room temperature as shown in figure 1(f). According to the previous reports, the peaks located below 200 cm⁻¹ reflect to low-frequency vibration and translation of tetrahedra-octahedra chains. The peaks at 300–500 cm^{-1} are associated with the deformation of tetrahedra and the translation of octahedra. The peaks at 600-800 cm^{-1} are ascribed with the stretching and bending of tetrahedra [52-54]. Notably, the Raman peak positions of our β -Ga₂O₃ microbelts are in good agreement with those for the previously reported bulk β -Ga₂O₃, suggesting the low defect density in β -Ga₂O₃ microbelts [52, 55, 56]. Figure 1(g) presents the high-resolution XPS O1s spectrum of β -Ga₂O₃ microbelts, which can be deconvoluted into two peaks centered at 530.68 eV (O_I) and 532.682 eV (O_{III}) . The peak at 530.68 eV can be assigned to the lattice oxygen in the β -Ga₂O₃ and the peak at 532.682 eV is generally considered to be the specific chemisorbed oxygen [57, 58]. Notably, the absence of the oxygen vacancy-related peak located near 531 eV (O_{II}) indicates that our β -Ga₂O₃ microbelts have extremely low oxygen vacancy defect density.

Figure 2(a) shows the *I–V* characteristic curves of the device in dark and under different illumination conditions. Under 20 V bias, the dark current of the device is only 1 pA, which is lower than that of the most reported Ga₂O₃ photodetectors [38, 39, 59–61]. Upon 254 nm illumination (650 μ W cm⁻²), the current shows a significant increase and the photo to dark current ratio can reach five orders of magnitude at 20 V. More interestingly, the device has an ultraweak response to 365 nm illumination (1050 μ W cm⁻²), suggesting an excellent UVA rejection. The *I–V* curves of the device under 365 and 254 nm illumination show a huge difference, which can be understood as follows: when the 254 nm



Figure 2. (a) *I*–*V* curves in dark and under 365 and 254 nm illuminations. (b) The spectral response property of the β -Ga₂O₃ solar-blind UV detector at a bias voltage of 20 V.

UV light is irradiated on the device, Ga₂O₃ absorbs light with photon energy greater than the semiconductor band gap, producing a large number of photo-generated carriers (electronhole pairs). As a result, the resistivity of Ga₂O₃ is reduced significantly. The photo-generated electrons and holes can be separated under the electric field, and then collected by the electrodes, which leads to the generation of large photocurrent. In contrast, when the 365 nm UV light is irradiated on the device, most photons pass through Ga2O3, and subbandgap defect-related absorption is very weak due to the high crystalline quality and few oxygen vacancy defects. Therefore, the device shows almost the same I-V curves both in dark and under 365 nm illumination. The photoresponse spectrum of the individual β -Ga₂O₃ microbelt photodetector is shown in figure 2(b). The device shows a peak responsivity of 9.47 A W⁻¹ at 256 nm with -3 dB cutoff wavelength of 264 nm, which is in good agreement with the bandgap of β -Ga₂O₃. More interestingly, the solar-blind/UV rejection ratio $(R_{\text{peak}}/R_{365 \text{ nm}})$ can reach as high as 8 \times 10⁴. In addition, the specific detectivity (D^*) is a critical parameter to describe the smallest detectable signal, which can be estimated by [62]

$$D^* = (S\Delta f)^{\frac{1}{2}} / \text{NEP}$$
(3)

where S is the effective illuminated area of the device, Δf is the bandwidth, and noise equivalent power (NEP) can be estimated by

$$NEP = I_n/R \tag{4}$$

where *R* is the responsivity, I_n is the noise current. For MSM solar-blind photodetectors, the noise is usually dominated by dark current shot noise, and thus I_n can be simply expressed as $(2qI_d\Delta f)^{\frac{1}{2}}$, where *q* is the electronic charge and I_d is the dark

current. So that, the D^* can be obtained by a more general equation [63–65]:

$$D^* = R/(2qI_{\rm d}/S)^{\frac{1}{2}}.$$
 (5)

The D^* value of the device at the wavelength of 256 nm was calculated to be as high as 1.67×10^{14} Jones at the bias of 20 V due to the high responsivity and low dark current.

Figure 3(a) shows the *I*-*t* curves of the device with different applied voltages under 254 nm illumination at a fixed power density of 650 μ W cm⁻². With increasing the voltage, the quick and stable on/off switching behavior can be well maintained. Meanwhile, the photocurrent increases linearly with the increase of bias voltage due to the enhanced drift velocity of the carriers (the inset). The time-dependent photoresponse measurement was carried out under different optical power densities at 20 V as shown in figure 3(b). And the photocurrent as a function of power density taken from the *I-t* curves was shown in the inset of figure 3(b). The nice linear relationship between photocurrent and optical power density can be clearly obtained, which is a strong and direct evidence of the high crystalline quality and low defect density of β -Ga₂O₃ microbelt, indicating the potential application of our device in light measurement and imaging [22, 66].

The response speed is another important figure of merit for UV photodetectors. To accurately obtain the response time of the device, the transient response was recorded using a 260 nm pulsed laser (10 ns pulse width and 10 Hz) and a digital oscilloscope as shown in figure 4. The 90%–10% fall time is only \sim 1.37 µs and the current can rapidly return to its initial dark-state value within 50 µs after the laser pulse.

To meet the requirements of practical applications, the mechanical and long-term stabilities of β -Ga₂O₃ microbelt solar-blind photodetector have been investigated. By bending the substrate at different angles, the mechanical stress can be



Figure 3. (a) *I*-*t* curves at different applied bias voltages under 254 nm light illumination with the intensity of 650 μ W cm⁻², and the inset is the photocurrent as a function of voltage. (b) *I*-*t* curves under 254 nm illumination with different intensities biased at 20 V, and the inset is the photocurrent as a function of illumination intensity.



Figure 4. The transient response of the device under 260 nm pulsed laser at 20 V.

introduced in the microbelt as shown in the inset of figure 5(a). At 20 V bias, the device exhibits almost constant performance under different bending states, suggesting an excellent mechanical stability (figure 5(a)). Moreover, the photoresponse behavior of β -Ga₂O₃ microbelt solar-blind photodetector remains unchanged after storing it under ambient environment for at least 3 months as presented in figure 5(b). This result indicates that the β -Ga₂O₃ microbelt-based photodetector possesses good long-term stability.

The excellent device performance in this work can be attributed to the high crystalline quality and few oxygen vacancies of the β -Ga₂O₃ microbelt, which is mainly realized by the following points: (a) compared with atmospheric pressure CVD, low pressure CVD provides a larger free diffusion distance of reactant gas molecules, a slower growth rate and higher purity, which is beneficial to obtain high crystalline quality β -Ga₂O₃ microbelts. (b) *In-situ* high-temperature post-annealing in Ar and O2 atmospheres can further improve the crystalline quality of the β -Ga₂O₃ microbelts and help to repair oxygen vacancy defects. Table 1 summarizes the comparison of the performance parameters of the recently reported 1D β -Ga₂O₃-based solar-blind UV photodetectors. Obviously, our single β -Ga₂O₃ microbelts solar-blind UV detector in this work has an ultrafast response speed with a decay time of only 1.37 μ s, which is more than three orders of magnitude faster than that of the previously reported 1D Ga₂O₃



Figure 5. Time–dependent photoresponse curves of the device (a) under different bending conditions and (b) after storage in air for 3 months.

Materials	Structure	Bias (V)	Dark current (A)	Responsivity (A W ⁻¹)	Solar-blind/UV rejection ratio	Decay time (s)	D* (Jones)	References
Ga ₂ O ₃ nanobelt	MSM	30	_	$\sim 10^2$	_	< 0.3	_	[29]
In-doped Ga ₂ O ₃ nanobelt	MSM	6	1×10^{-13}	547		<0.6		[34]
Ga ₂ O ₃ multi-nanobelt	MSM	20	1×10^{-13}	851		<0.3		[37]
Ga ₂ O ₃ /PANI	p–n	0	_	$2.1 imes 10^{-2}$	_	8.14×10^{-3}	1.5×10^{11}	[40]
Ga ₂ O ₃ micro/nano-sheet	Schottky	1	$\sim 10^{-10}$	19.31	_	0.023	_	[38]
β -Ga ₂ O ₃ nanowires	MSM	6	3.25×10^{-10}	0.128	46.28 $(R_{254 \text{ nm}}/R_{365 \text{ nm}})$	3.188	3.676×10^{10}	[41]
β -Ga ₂ O ₃ nanowires	MSM	5		0.122		0.012	_	[39]
β -Ga ₂ O ₃ /GaN nanowires	Heterojunction	10	1.6×10^{-10}	2.75×10^{-2}	_	0.23	1.2×10^{11}	[42]
β -Ga ₂ O ₃ nanowires	MSM	10	7.73×10^{-10}	233	_	0.04	8.16×10^{12}	[43]
β -Ga ₂ O ₃ nanowires	MSM	20	$8 imes 10^{-9}$	0.71	_	0.19	_	[44]
Ga ₂ O ₃ /GaN nanowire array	Vertical heterojunction	5	_	753.2	_	<0.04	_	[36]
β -Ga ₂ O ₃ nanowires	MSM	10	8×10^{-12}	—	$\sim 10^3$ (<i>R</i> _{231 nm} / <i>R</i> _{290 nm})	<0.5	_	[45]
Ga ₂ O ₃	MSM	20	1×10^{-12}	9.47	$(R_{254 \text{ nm}}/R_{365 \text{ nm}})$ $(R_{254 \text{ nm}}/R_{365 \text{ nm}})$	1.37×10^{-6}	1.67×10^{14}	This work

Table 1.	Performance compar	ison of representa	ive 1D β -Ga ₂ (D_3 based	l solar-blind U	JV detectors.
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devices. Moreover, owing to the ultralow dark current and relatively high responsivity, our device exhibits the highest specific detectivity of $\sim 10^{14}$ Jones and the highest solar-blind/UV rejection ratio of nearly 10^5 , among the photodetectors based on Ga_2O_3 1D structures reported so far. The devices with high solar blind/UV rejection ratio and high-speed have important application prospects in practical applications high frequency and weak signal detection.

4. Conclusions

In this work, the high-quality β -Ga₂O₃ microbelts were prepared through low-pressure CVD and *in-situ* atmospheric pressure annealing technologies. Solar-blind UV detector with MSM structure was fabricated using a single β -Ga₂O₃ microbelt with two gold electrodes spaced 100 μ m apart. The device exhibits a low dark current of $\sim 10^{-12}$ A, a peak responsivity of 9.47 A W⁻¹ at 256 nm and an ultrashort decay time of 1.37 μ s at the bias of 20 V. Moreover, the specific detectivity of $\sim 10^{14}$ Jones and solar-blind/UVA rejection ratio $(R_{\text{peak}}/R_{365 \text{ nm}})$ of nearly 10⁵ have also been achieved in this work, which are much higher than those of the previously reported Ga₂O₃ 1D micro/nanostructure photodetectors. Besides that, β -Ga₂O₃ microbelt photodetector exhibits excellent mechanical and long-term stabilities. Our findings provide a feasible method to develop high performance, high stable and flexible solar-blind UV photodetectors.

Data availability statement

All data that support the findings of this study are included within the article (and any supplementary files).

Acknowledgments

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Author contributions

The manuscript was written through contributions of all authors. Zhiyao Zheng: Investigation, Writing-original draft. Kewei Liu: Validation, Formal analysis, Writing-review and editing, Supervision, Resources. Zhen Cheng, Resources, Visualization. Baoshi Qiao, TEM analyse. Xing Chen, Resources, Visualization. Chang Zhou, Investigation. Jialin Yang: Supervision. Qiu Ai, Supervision. Yongxue Zhu, Supervision. Binghui Li: Resources, Supervision. Lei Liu: Resources, Supervision. Dezhen Shen: Writing-review and editing, Supervision, Resources. All authors have given approval to the final version of the manuscript.

Conflict of interest

There are no conflicts to declare.

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References

- [1] Chang M et al 2022 J. Phys. D: Appl. Phys. 55 035103
- [2] Liu Z, Zhang S, Zhi Y, Li S, Yan Z, Chu X, Bian A, Li P and Tang W 2021 J. Phys. D: Appl. Phys. 54 195104
- [3] Varshney U, Aggarwal N and Gupta G 2022 J. Mater. Chem. C 10 1573
- [4] Li S et al 2021 J. Phys. Chem. Lett. 12 447
- [5] Xie C, Lu X, Liang Y, Chen H, Wang L, Wu C, Wu D, Yang W and Luo L 2021 J. Mater. Sci. Technol. 72 189
- [6] Xie C, Lu X-T, Tong X-W, Zhang Z-X, Liang F-X, Liang L, Luo L-B and Wu Y-C 2019 Adv. Funct. Mater. 29 1806006
- [7] Xie C and Yan F 2017 Small 13 1701822
- [8] Li S et al 2019 ACS Appl. Mater. Interfaces 11 35105
- [9] Li Y, Xiu X, Xu W, Zhang L, Xie Z, Tao T, Chen P, Liu B, Zhang R and Zheng Y 2021 J. Phys. D: Appl. Phys. 54 014003
- [10] Ma Y et al 2021 J. Phys. D: Appl. Phys. 54 305101
- [11] Cai Q, You H, Guo H, Wang J, Liu B, Xie Z, Chen D, Lu H, Zheng Y and Zhang R 2021 Light Sci. Appl. 10 94
- [12] Chen Y, Zhou X, Zhang Z, Miao G, Jiang H, Li Z and Song H 2021 Mater. Lett. 291 129583
- [13] Han D, Liu K, Chen X, Li B, Zhai T, Liu L and Shen D 2021 Appl. Phys. Lett. 118 251101
- [14] Wang Z, Lin J, Wei X, Zheng W and Hu Q 2021 Front. Mater. 8 787613
- [15] Chen X, Wang L, Liu K, Zhang Z, Li B, Wu J, Wang J, Ni Y and Shen D 2020 J. Mater. Chem. C 8 1089
- [16] Tsai S H, Basu S, Huang C Y, Hsu L C, Lin Y G and Horng R H 2018 Sci. Rep. 8 14056
- [17] Han D, Liu K, Hou Q, Chen X, Yang J, Li B, Zhang Z, Liu L and Shen D 2020 Sens. Actuators A 315 112354
- [18] Liu Z, Ao J-P, Li F, Wang W, Wang J, Zhang J and Wang H-X 2016 Appl. Phys. Lett. **109** 153507
- [19] Liang Y, Ma M, Zhong X, Xie C, Tong X, Xing K and Wu C 2021 IEEE Electron Device Lett. 42 1192
- [20] Ma Y J et al 2022 Vacuum 198 110886
- [21] Tang X et al 2022 ACS Appl. Mater. Interfaces 14 1304
- [22] Tak B R, Kumar S, Kapoor A K, Wang D, Li X, Sun H and Singh R 2021 J. Phys. D: Appl. Phys. 54 453002
- [23] Zheng Y, Hasan M N and Seo J H 2021 Adv. Mater. Technol. 6 2100254
- [24] Qiao B, Zhang Z, Xie X, Li B, Li K, Chen X, Zhao H, Liu K, Liu L and Shen D 2019 J. Phys. Chem. C 123 18516
- [25] Feng Q et al 2016 IEEE Trans. Electron Devices 63 3578
- [26] Khan M, Kadam V, Sant T, Jejurikar S M, Joshi K, Bhopale S R, More M A and Late D 2020 Mater. Chem. Phys. 254 123512
- [27] Wang Y H et al 2021 ACS Nano 15 16654
- [28] Pratiyush A S, Krishnamoorthy S, Solanke S V, Xia Z, Muralidharan R, Rajan S and Nath D N 2017 Appl. Phys. Lett. 110 221107
- [29] Li L, Auer E, Liao M, Fang X, Zhai T, Gautam U K, Lugstein A, Koide Y, Bando Y and Golberg D 2011 Nanoscale 3 1120
- [30] Chen M, Zhang Z, Lv Z, Zhan R, Chen H, Jiang H and Chen J 2022 ACS Appl. Nano Mater. 5 351
- [31] Li Z M et al 2020 Vacuum 178 109448

- [32] Nishinaka H, Nagaoka T, Kajita Y and Yoshimoto M 2021 Mater. Sci. Semicond. Proc. 128 105732
- [33] Lian Q et al 2019 Small 15 1900236
- [34] Tian W, Zhi C, Zhai T, Chen S, Wang X, Liao M, Golberg D and Bando Y 2012 J. Mater. Chem. 22 17984
- [35] Li S X, Xu Y S, Li C L, Guo Q, Wang G, Xia H, Fang H H, Shen L and Sun H B 2020 Adv. Mater. 32 2001998
- [36] He T et al 2019 Adv. Opt. Mater. 7 1801563
- [37] Zou R, Zhang Z, Liu Q, Hu J, Sang L, Liao M and Zhang W 2014 Small 10 1848
- [38] Oh S, Mastro M A, Tadjer M J and Kim J 2017 ECS J. Solid State Sci. 6 Q79
- [39] Zhang L et al 2020 Nanophotonics 9 4497
- [40] Wang Y, Li L, Wang H, Su L, Chen H, Bian W, Ma J, Li B, Liu Z and Shen A 2020 Nanoscale 12 1406
- [41] Zhang M, Kang S, Wang L, Zhang K, Wu Y, Feng S and Lu W 2021 J. Phys. D: Appl. Phys. 54 175106
- [42] Ding W and Meng X 2021 J. Alloys Compd. 866 157564
- [43] Xie C, Lu X T, Ma M R, Tong X W, Zhang Z X, Wang L, Wu C Y, Yang W H and Luo L B 2019 Adv. Opt. Mater. 7 1901257
- [44] Wang S et al 2019 J. Alloys Compd. 787 133
- [45] Du J, Xing J, Ge C, Liu H, Liu P, Hao H, Dong J, Zheng Z and Gao H 2016 J. Phys. D: Appl. Phys. 49 425105
- [46] Zhang Z G, Wang X X, Zhang J, Yu M, Zhang J C, Zhang H D and Long Y Z 2018 J. Alloys Compd. 752 359
- [47] Oh S, Kim J, Ren F, Pearton S J and Kim J 2016 J. Mater. Chem. C 4 9245
- [48] Kim J, Oh S, Kim S and Kim J 2019 Ga₂O₃ nanobelt devices Gallium Oxide (Amsterdam: Elsevier) p 331
- [49] Ohira S, Sugawara T, Nakajima K and Shishido T 2005 J. Alloys Compd. 402 204

- [50] Xiang H C, Han S, Lu Y M, Cao P J, Liu W J, Zeng Y X, Jia F, Xu W Y, Liu X K and Zhu D L 2018 J. Alloys Compd. 747 869
- [51] Wang D, He L, Le Y, Feng X, Luan C, Xiao H and Ma J 2020 Ceram. Int. 46 4568
- [52] Dohy D, Lucazeau G and Revcolevschi A 1982 J. Solid State Chem. 45 180
- [53] Onuma T, Fujioka S, Yamaguchi T, Itoh Y, Higashiwaki M, Sasaki K, Masui T and Honda T 2014 J. Cryst. Growth 401 330
- [54] Kumar S, Dhara S, Agarwal R and Singh R 2016 J. Alloys Compd. 683 143
- [55] Zhao Y and Frost R L 2008 J. Raman Spectrosc. 39 1494
- [56] Kranert C, Sturm C, Schmidt-Grund R and Grundmann M 2016 Sci. Rep. 6 35964
- [57] Liu S, Jiao S, Lu H, Nie Y, Gao S, Wang D, Wang J and Zhao L 2022 J. Alloys Compd. 890 161827
- [58] Weng T F, Ho M S, Sivakumar C, Balraj B and Chung P F 2020 Appl. Surf. Sci. 533 147476
- [59] Cao J, Chen L, Chen X, Zhu Y, Dong J, Wang B, He M and Wang X 2021 Crystals 11 1248
- [60] Tan P et al 2021 Adv. Opt. Mater. 9 2100173
- [61] Zhong M, Wei Z, Meng X, Wu F and Li J 2015 J. Alloys Compd. 619 572
- [62] Fang Y, Armin A, Meredith P and Huang J 2018 Nat. Photon.13 1
- [63] Kim S and Kim J 2020 Appl. Phys. Lett. 117 261101
- [64] Wang Y et al 2019 Appl. Phys. Lett. 114 011103
- [65] Oh S, Kim C-K and Kim J 2017 ACS Photonics 5 1123
- [66] Zheng Z, Liu K, Chen X, Qiao B, Ma H, Liu D, Liu L and Shen D 2021 Nanotechnology 32 475201