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Highly efficient organic tandem solar cell with a SubPc interlayer based on TAPC:C₇₀ bulk heterojunction

Yuan Gao^{1,2}, Fangming Jin^{1,2}, Wenlian Li¹, Zisheng Su¹, Bei Chu¹, Junbo Wang¹, Haifeng Zhao¹, Hairuo Wu^{1,2}, Chengyuan Liu^{1,2}, Fuhua Hou^{1,2}, Tong Lin^{1,2} & Qiaogang Song^{1,2}

We report a small molecule tandem organic photovoltaic (OPV) cell with a high power conversion efficiency (PCE) of 7.27%. This cell contains two subcells with an identical mixed active layer of C_{70} :5 wt%TAPC (1,1-bis-(4-bis(4-methyl-phenyl)-amino-phenyl)-cyclohexane). The performance was dramatically improved by simply inserting a thin boron subphthalocyanine chloride (SubPc) interlayer, which results in an increase of the short-circuit current and open-circuit voltage as well as a decrease of the series resistance of the tandem cell. The response of the cell only contributed from the absorption of C_{70} . The high PCE was attributed to the high absorption efficiency of C_{70} and improved holes extraction efficiency at the anode due to the band bending occurs at both MoO₃/SubPc and SubPc/C₇₀:5 wt%TAPC interfaces.

Over the past few years, organic photovoltaic (OPV) cells have drawn a great deal of attention in both fundamental research and industrial fields. OPV cell is one of the most-promising next-generation energy-harvesting technologies because it has the advantages of low weight, flexible form factor, low-cost mass production and low energy consumption in manufacturing processes¹⁻⁸. Recently, the power conversion efficiency (PCE) of the OPV cells has exceeded 10%, which is a milestone development for valuable commercial applications^{9,10}. In spite of this, there are some factors limit the performance of OPV cells such as low light absorption and charge carrier collection efficiencies. A tandem cell architecture is usually adopted to increase the absorption and hence the performance of OPV cells. Besides, the architecture of tandem cells should be precisely designed to guarantee a high charge carrier collection efficiency. Typically, tandem cells are constructed by series stacking two or more subcells with complementary absorption spectra because the overall light absorption could be enhanced. As a result of such a design, a matched photocurrent resulted from the individual subcell was extracted¹¹. The subcells with identical compositions have also been used in tandem structure, but the PCE of these cells was generally lower^{12,13}. Zhang *et al.*¹³ reported a tandem cell with the same subcell units of MoO_3/C_{60} :5 wt% 1,1-bis-(4-bis(4-methyl-phenyl)-amino-phenyl)-cyclohexane (TAPC). This subcell structure was similar to their previous reported OPV cells¹⁴, in which the open circuit voltage (Voc) was mainly determined by the MoO₃/C₆₀:TAPC Schottky barrier¹⁵ and the short circuit current (Jsc) was primary originated from the absorption of the fullerene. This tandem OPV cells had a lower PCE of 4.12%. This should be attributed the low absorption efficiency of C_{60} and low charge carrier collection efficiency of this device¹³. In one of our previous works, we have observed a considerable improved PCE of the OPV cell when a thin boron subphthalocyanine chloride (SubPc) interlayer was inserted between ITO/MoO₃ and bulk heterojunction (BHJ) of TAPC: C_{60}^{16} . A high hole extraction efficiency has been obtained in planar heterojunction (PHJ) OPV cells with SubPc and fullerene materials as the donor and acceptor, respectively^{17,18}. Because SubPc has a deep highest occupied molecular orbital (HOMO) level (i.e., higher ionization potential), a significant band bending towards the gap states in the MoO₃ layer¹⁸ due to interface dipole effect can be observed at the MoO₃/SubPc interface^{17,19}. Such an energy level alignment allows more efficient holes extraction of the devices¹⁷. C₇₀ fullerene has comparable electronic properties but a higher extinction

¹State Key Laboratory of Luminescence and Applications, Changchun Institute of Optics, Fine Mechanics, and Physics, Chinese Academy of Sciences, Changchun 130033, People's Republic of China. ²University of Chinese Academy of Sciences, Beijing 100039, People's Republic of China. Correspondence and requests for materials should be addressed to W.L. (email: wllioel@aliyun.com) or Z.S. (email: suzs@ciomp.ac.cn)



Figure 1. Device architecture of the single cell and (b) tandem cell with changed thickness ratio between front and back subcell.



Figure 2. The J-V curves of single cells with different thickness of SubPc.

coefficient at low energy wavelength region as compared with C_{60} fullerene. This leads to a higher PCE of OPV cells with a C_{70} acceptor.

In this work, tandem OPV cells with consistent subcell that comprising a C_{70} :5 wt%TAPC active layer is designed. By inserting a thin SubPc layer between MoO₃ and the TAPC: C_{70} BHJ layer, a high PCE of 7.27% is obtained. In light of the above descriptions, our new tandem cell system exist two MoO₃-SubPc Schottky junctions which lead to a stronger built-in field in the cell due to a thinner SubPc and a BHJ active layer were adopted in the subcells. As a result of such a structure, we can expect a high charge carrier extraction efficiency and hence a high PCE of the tandem cell. Our optimized tandem cell was addressed by series stacking the front- and back-cells, which has an architecture of ITO/MoO₃ (5 nm)/SubPc (1 nm)/5 wt%TAPC: C_{70} (33 nm)/Bphen (2 nm)/Ag (0.5 nm)/MoO₃ (5 nm)/SubPc (1 nm)/5 wt%TAPC: C_{70} (33 nm)/Bphen (2 nm)/Ag (0.5 nm)/MoO₃ (5 nm)/SubPc (1 nm)/5 wt%TAPC: C_{70} (34 nm)/Al (100 nm). In the end, a peak PCE of 7.27% was successfully demonstrated, which is one of the highest value among the reported tandem OPV cells with two identical subcells. It is also interesting that the increase in spectral response of our new single cell mainly focuses on the absorption of C_{70} acceptor. That is, there was no complementary absorption in our new tandem cell, which is in contrast to conventional reported tandem cells²⁰. Besides, the donor and acceptor materials used here are all conventional and extensive materials. Therefore, we believe that cost-efficient OPV cells can be developed thanks to the design of the tandem cell architecture and the utilization of facile materials, for example, C_{70} , SubPc, TAPC, Ag, and so on.

Results and Discussion

Prior to constructing the tandem OPV cells, a series of single cells with the structure of ITO/MoO_3 (5 nm)/SubPc (x nm)/C₇₀:5 wt%TAPC (60 nm)/Bphen (8 nm)/Al (100 nm) were constructed, here x ranged from 0.5 to 2.0 nm, as shown in Fig. 1(a). The J–V curves of these cells are presented in Fig. 2, and the PV parameters extracted from these curves are listed in Table 1. Each curve is averaged over four devices with the same configuration. The reference cell without SubPc layer shows a Jsc, Voc, fill factor (FF), and PCE of 8.94 mA/cm², 0.85 V, 0.51, and 3.88%, respectively. After inserting a thin SubPc layer, the performance of the cells is significantly improved. The optimized cell with 1 nm SubPc shows a Jsc, Voc, FF, and PCE of 10.98 mA/cm², 0.85 V, 0.53, and 4.95%, respectively. The PCE is increased by 27.6% compared with the reference cell which is mainly ascribed to the increased Jsc

Thickness (nm)	Jsc(mA/cm ²)	Voc(V)	FF	PCE(%)	R _s (ohm.cm ²)
0.0	8.94 ± 0.07	0.85 ± 0.01	0.51 ± 0.01	3.88 ± 0.12	7.73
0.5	10.40 ± 0.02	0.86 ± 0.01	0.52 ± 0.02	4.63 ± 0.27	5.79
1.0	10.98 ± 0.01	0.85 ± 0.01	0.53 ± 0.01	4.95 ± 0.06	2.69
1.5	9.55 ± 0.19	0.84 ± 0.01	0.52 ± 0.01	4.23 ± 0.01	3.54
2.0	10.46 ± 0.21	0.79 ± 0.01	0.45 ± 0.01	3.75 ± 0.18	9.67

Table 1. PV parameters and series resistance (Rs) of single cell with different thickness of SubPc.

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Figure 3. The external quantum efficiency (EQE) spectra of single cells with different thickness of SubPc films.





and FF. Meanwhile, a decrease of series resistance is also observed (see Table 1), which implies that there is a low carrier recombination probability in these cells.

The EQE spectra of the single OPV cells with different thickness of SubPc are showed in Fig. 3. All these cells show a similar shape of EQE spectra with two bands located at about 400 and 550 nm, respectively. Besides, we can find that the cell with a 1 nm SubPc exhibits the highest EQE in the whole visible region, which is consistent with the highest Jsc observed in the J-V curves. The calculated Jsc from this EQE spectrum is 10.71 mA/cm², which is a little lower than that obtained from the J-V curve but still within the range of allowable error. The normalized absorption spectra of 50 nm SubPc, C₇₀, and TAPC films are depicted in Fig. 4, as well as their chemical structures. C₇₀ has two absorption bands at about 390 and 520 nm, while the absorption band of SubPc locates at about 590 nm. The EQE spectra are more consistent to the absorption spectrum of C_{70} , indicating that the photocurrent is predominantly contributed from the absorption of C70. To further distinguish the contributions of the absorptions of SubPc and C₇₀ to the photocurrent, the absorption spectra of SubPc/C₇₀:TAPC films with different SubPc thickness are investigated, as shown in Supplementary Fig. S1. The absorption intensity at about 590 nm which comes from SubPc slightly increases with the thickness of SubPc. However, no obvious respond at 590 nm is found in the EQE spectra regardless of its thickness. The J-V curves and the response of the devices $ITO/MoO_3/SubPc/C_{70}/Bphen/Al$ are shown in Supplementary Fig. S2. The response of the devices exhibit strong peaks at about 590 nm, which can be assigned to the absorption of SubPc. These findings suggest that SubPc does not act as a donor but more likely only as an interlayer in these cells.

Thickness (x, y nm)	Jsc(mA/cm ²)	Voc(V)	FF	PCE(%)	R _s (ohm.cm ²)
(24.5, 49)	6.19 ± 0.16	1.57 ± 0.01	0.56 ± 0.01	5.42 ± 0.16	11.19
(25, 49)	6.27 ± 0.15	1.58 ± 0.04	0.57 ± 0.03	5.63 ± 0.37	13.41
(25, 54)	7.57 ± 0.35	1.58 ± 0.01	0.48 ± 0.01	5.79 ± 0.33	42.54
(28, 54)	6.80 ± 0.17	1.59 ± 0.01	0.57 ± 0.03	6.14 ± 0.16	19.00
(28, 59)	6.69 ± 0.19	1.50 ± 0.01	0.56 ± 0.01	5.63 ± 0.19	9.30
(28, 64)	7.09 ± 0.07	1.56 ± 0.01	0.41 ± 0.01	4.80 ± 0.23	15.40
(28, 74)	5.47 ± 0.34	1.51 ± 0.03	0.39 ± 0.05	3.26 ± 0.43	43.36
(33, 54)	8.70 ± 0.23	1.63 ± 0.01	0.51 ± 0.01	7.27 ± 0.18	42.69
Single cell	10.98 ± 0.01	0.85 ± 0.01	0.53 ± 0.01	4.95 ± 0.06	2.69

Table 2. The PV parameters of a series of tandem cells with different thickness of active layer of C_{70} :5 wt%TAPC, respectively.



Figure 5. The relation between short curve current density and voltage of the optimized tandem and the corresponding reference single cell with different thickness of active layer.

Based on these results, we fabricated a series of tandem cells with the structures of ITO/[MoO₃ (5 nm)/SubPc (1 nm)/ C_{70} :5 wt%TAPC (x nm)/Bphen (2 nm)](S1)/Ag (0.5 nm)/[MoO₃ (5 nm)/SubPc (1 nm)/ C_{70} :5 wt%TAPC (y nm)/Bphen (8 nm)](S2)/Al (100 nm), as depicted in Fig 1(b), here x and y represents the thickness of the active layer, respectively. We can see that the two subcells (S1 and S2) are connected by a 0.5 nm Ag layer which acts as the recombination centre of electrons and holes that originated from S1 and S2, respectively. The transmittance spectra of 0.5 nm Ag and Bphen (2 nm)/Ag (0.5 nm)/MoO₃ (5 nm) are showed in Supplementary Fig. S3. It can be seen that the transmittances are more than 94% in the range from 300 to 800 nm, ensuring enough light to be absorbed by the top cell. Hence, this intermediate layer satisfies the optical requirement for device fabricated.

The PV parameters of the tandem cells with different thickness of front and back subcells are shown in Table 2. Figure 5 compares the J–V curves of the optimized tandem cell and reference single cells. The single cells with 33, 54, and 60 nm C_{70} :5 wt% TAPC layers shows a PCE of 3.49, 4.02, and 4.95, respectively, while the optimized tandem cell with 33 and 54 nm C_{70} :5 wt% TAPC in S1 and S2, respectively has a maximum PCE of 7.27%. This suggests that the PCE of the tandem cell is about 47% higher than that of the reference single one. We also note that the Voc of the tandem cell is almost double that of the single cell, which indicates that our designed tandem OPV cells are in accordance with the requirements of the tandem OPV cells to improve their performance^{13,21,22}. It is surprising that the FF is almost unchanged for the tandem and single cells due to their lower series resistance, as shown in Table 2. As a result, a higher Jsc of 8.70 mA/cm² and a FF of 0.51 are simultaneously obtained. More interesting, all the parameters of the tandem cells have a low standard deviation, indicating that a high reproducibility of these tandem cells.

To further understand the effects of the SubPc layer on the performance of the OPV cells, AFM images of SubPc with different thickness on 5 nm MoO₃ were investigated, as presented in Fig. 6(a-e). The root-mean-square (RMS) roughness of pristine MoO₃ layer is 0.478 nm and then changes to be 0.420, 0.341, 0.448 and 0.373 nm for 0.5, 1.0 nm, 1.5 nm and 2.0 nm SubPc on MoO₃, respectively. This difference RMS roughness between the films indicates that 1.0 nm SubPc covers majority of MoO₃ surface, while 0.5 nm SubPc has a lower coverage on MoO₃ surface. The increase of RMS roughness with 1.5 nm SubPc should be attributed to the aggregation of the ultra thin SubPc layer, while it will form a uniform film when the thickness of SubPc increases to 2.0 nm due to the crystallization of SubPc, which resultes in decease of RMS roughness. In spite of this, the cell with 2 nm SubPc presents an increased series resistance, thus the FF and hence the PCE is decreased (See Fig. 2 and Table 1). The difference between the morphologies also corresponds to the EQE spectra of the cells with different thickness of SubPc, as shown in Fig. 3.

The energy level alignment of the cells plays an important role in determining the charge carrier extraction efficiency. Thus we investigate the energy level alignment at C_{70} /SubPc/MoO₃/ITO using UPS, as indicated in



Figure 6. AFM images $(1 \ \mu m \times 1 \ \mu m)$ of (**a**) neat 5 nm thin film of MoO₃ on Si substrate. (**b**-**e**) on Si substrate, neat 5 nm thin film of MoO₃ coated with various SubPc thin film thickness (0.5, 1.0, 1.5 and 2 nm respectively). The root-mean-square (rms) roughness of the image from (**a**-**e**) were 0.478, 0.420, 0.341, 0.448 and 0.373 respectively.





Fig. 7. As a result of the band bending in both TAPC and C_{60} layer deposited on MoO₃, a built-in field would be resided²³. Here we only used pure C_{70} layer rather than the C_{70} :5 wt% TAPC layer in the experiment. From Fig. 7, we note that there is an interface dipole of $\Delta_1 = 0.45$ eV at the MoO₃/ITO interface, and this interface can not be changed by the subsequent deposition of SubPc¹⁹ as it only affects the surface state of MoO₃. The vacuum level (VL), and valence band (VB) of MoO₃ were measured to be 5.25, and 2.75 eV, respectively. A dipole of $\Delta_2 = 0.25$ eV is formed at the SubPc/MoO₃ interface, inducing a behaviour of band bending in SubPc. The band-bending phenomenon in SubPc layer presents a drift electric field inside the organic material^{17,19}. At the very interface of SubPc/MoO₃, the VL, LUMO and HOMO of SubPc are 5.00, 1.42, and 0.68 eV, then they change to 4.88, 1.36, and 0.74 eV, respectively, with the increase of the thickness of SubPc. With the subsequent deposition of C_{70} , the VL, LUMO, and HOMO of C_{70} are read as 5.13, 1.73, and 0.7 eV, respectively. A gradual energy level shift is observed with the further C_{70} deposition until to 5 nm, indicating exists a band bending region. So after C_{70} :5 wt%TAPC mixed layer was deposited on SubPc/MoO₃/ITO, a band bending in both SubPc and C_{70} : TAPC BHJ towards MOO₃ would take place²³. The more detail analysis on these interfaces by UPS experiments would be



Figure 8. The schematic level alignment and band bending as well as the exciton dissociation, carrier transporting route and recombination in Ag ICL. White and black circles denote hole and electrons, bending red arrows show the carrier transport direction, bending green, blue and organic lines denote energy band bending of SubPc, TAPC and C₇₀, respectively. The closer hole and electron lying in the yellow Ag ICL denote their recombination.

discovered further. In view of these, the holes from C_{70} :TAPC-BHJ can be easily transported to the anode via the bending HOMO level of SubPc due to the reduced holes extraction barrier.

Thus the work mechanism of the tandem OPV cells can be explained as follows. Firstly, when the incident light was entered into such a tandem cell, the excitons are generated in the C_{70} :5 wt%TAPC active layer of both the subcells due to the absorption of C_{70} . The excitons of C_{70} can be quickly dissociated into free holes and electrons by the HOMO energy level offset between TAPC and C_{70}^{24} and aforementioned built-in field induced by both band bendings. Besides, the excitons of C_{70} can also be dissociated at the SubPc/ C_{70} interface due to the HOMO energy level offset between these two materials, as shown in Fig. 7. Because the favorable energy level alignment at the MoO₃/SubPc and SubPc/ C_{70} interfaces can improve exciton dissociation and decrease the potential barrier for charge extraction, high Jsc and PCE were harvested. Secondly, the thinner total thickness of the two subcells in the tandem cell also possesses another advantage, i.e., the illumination light could enter the back subcell more easily. As a result, a balanced photocurrent would generate in the two subcells. Finally, after the excitons dissociate in the two subcells, the working processes and mechanisms are showed in Fig. 8, where the schematic level alignment, band bending, the exciton dissociation, carrier transporting route and recombination in Ag intermediate layer are clearly displayed. There are two built-in fields in the cell, one is arising from MoO₃-SubPc Schottky-junction^{13,17} and the other one is from the offset between HOMO levels of C₇₀ and TAPC. Under the driving of the built-in fileds in the front subcell, the photogenerated holes are transported to the anode via the HOMOs of TAPC and SubPc, while the electrons are transported to Ag intermediate layer through the LUMO of C70 and the defect states of Bphen. Similar situations will occur in the back subcell, but with the holes transported to Ag intermediate layer and electrons to Al cathode. Then recombination between holes and electrons takes place at Ag nanoparticles. Furthermore, it is also remarkable that the very thin SubPc interlayer not only induces band bending of C₇₀:5 wt%TAPC layer but also prevents exciton quenching by MoO₃. Exciton quenching at the MoO₃/ organic interface plays an important role in determining the PCE of the OPV cells, especially in the cells where the BHJ active layer with a relatively high concentration of C_{70} is directly contacted with MoO₃²⁵. In a word, the introduction of a SubPc organic layer in our tandem cell does improve the performance of the tandem cells and a more detailed working mechanism is still under research.

Conclusion

We have demonstrated an efficient tandem OPV cell with identical subcells based on C_{70} :5 wt%TAPC. By inserting an ultrathin SubPc layer, a PCE as high as 7.27% was demonstrated. The response of the cell primary contributed from the absorption of C_{70} . The high efficiency was attributed to the high absorption efficiency of C_{70} and improved holes extraction efficiency at the anode due to the band bending occurs at the MoO₃/SubPc and SubPc/ C_{70} :5 wt%TAPC interfaces. These findings indicate that high PCE tandem OPV cells with identical subcells can be constructed by precise device design, and this work provides a new strategy to fabricate high performance tandem OPV cells.

Methods

Fabrication of small molecule materials based organic photovoltaic cells. The organic materials used for all the OPV cells were procured commercially and were used without further sublimation. The ITO-coated glass substrates with a sheet resistance of $15 \Omega/sq$ were routinely cleaned in a series of solvents and then treated by ultraviolet-ozone in a chamber for $15 \min$. Organics and metal layers were sequentially deposited onto the clean patterned ITO substrates by thermal evaporation in a vacuum chamber at a pressure of 4×10^{-4} Pa without a vacuum break. The area of the devices patterned by the shadow mask was 0.1 cm^2 . The deposition rates were monitored using quartz oscillating crystals and kept to be 0.5 Å/s for organic layer and 5 Å/s for the Al cathode.

Device measurement and characterization. Absorption spectra of the organic films on quartz substrates were measured with a Shimadzu UV-3101PC spectrophotometer. The current density-voltage (J–V) characteristics of the OPV cells were measured using a computer-controlled Keithley 2611 source meter under AM 1.5 G illumination from a calibrated Solar with an irradiation intensity of 100 mW·cm⁻². The external quantum efficiency (EQE) measurements were performed with a lock-in amplifier at a chopping frequency of 20 Hz during illumination with monochromatic light from a xenon lamp and their intensities are calibrated with a Si-photodiode. The surface topographies were imaged with a Bruker MultiMode 8 atomic force microscope (AFM) in tapping mode. The UPS experiments were performed using a VG ESCA Lab system equipped with a He I (21.22 eV) gas discharge lamp. The ultrahigh vacuum (UHV) system consists of a spectrometer chamber and an evaporation chamber. The base pressures of the spectrometer chamber and the evaporation chamber are typically 1.1×10^{-8} Pa and 6.7×10^{-6} , respectively. We recorded the UPS spectra with the samples biased at -4.0 V to observe the true low energy secondary cut-off. The UV light spot size on samples was about 1 mm in diameter. The instrumental resolution for UPS measurements was chosen to be 10 meV. All the measurements were carried out at room temperature.

References

- 1. Peumans, P., Uchida, S. & Forrest, S. R. Efficient bulk heterojunction photovoltaic cells using small-molecular-weight organic thin films. *Nature* 425, 158–162 (2003).
- 2. Uchida, S., Xue, J., Rand, B. P. & Forrest, S. R. Organic small molecule solar cells with a homogeneously mixed copper phthalocyanine: C₆₀ active layer. *Appl. Phys. Lett.* 84, 4218–4220 (2004).
- Yang, F., Shtein, M. & Forrest, S. R. Controlled growth of a molecular bulk heterojunction photovoltaic cell. Nat Mater 4, 37–41 (2005).
- 4. Günes, S., Neugebauer, H. & Saricifici, N. S. Conjugated polymer-based organic solar cells. Chem Rev 107, 1324-1338 (2007).
- Rand, B. P., Genoe, J., Heremans, P. & Poortmans, J. Solar cells utilizing small molecular weight organic semiconductors. *Prog. Photovolt: Res Appl.* 15, 659–676 (2007).
 Chen, Y.-H. *et al.* Vacuum-Deposited Small-Molecule Organic Solar Cells with High Power Conversion Efficiencies by Judicious
- 6. Chen, 1.-H. et al. Vacuum-Deposited smain-Molecule Organic Solar Cens with Figh Power Conversion Enciencies by Judicious Molecular Design and Device Optimization. J. Am. Chem. Soc. 134, 13616–13623 (2012).
- 7. You, J. et al. A polymer tandem solar cell with 10.6% power conversion efficiency. Nat. Commun. 4, 1446 (2013).
- Cao, W. & Xue, J. Recent progress in organic photovoltaics: device architecture and optical design. Energ. Environ. Sci. 7, 2123–2144 (2014).
- 9. Dennler, G. *et al.* Design Rules for Donors in Bulk-Heterojunction Tandem Solar Cells Towards 15% Energy-Conversion Efficiency. *Adv. Mater.* **20**, 579–583 (2008).
- 10. Ameri, T., Dennler, G., Lungenschmied, C. & Brabec, C. J. Organic tandem solar cells: a review. Energ. *Environ. Sci.* 2, 347–363 (2009).
- Li, K. et al. Development of large band-gap conjugated copolymers for efficient regular single and tandem organic solar cells. J. Am. Chem. Soc. 135, 13549–13557 (2013).
- 12. Kawano, K., Ito, N., Nishimori, T. & Sakai, J. Open circuit voltage of stacked bulk heterojunction organic solar cells. *Appl. Phys. Lett.* **88**, 073514 (2006).
- Zhang, M., Wang, H. & Tang, C. W. Tandem photovoltaic cells based on low-concentration donor doped C₆₀. Org. Electron. 13, 249–251 (2012).
- 14. Zhang, M., Wang, H., Tian, H., Geng, Y. & Tang, C. W. Bulk heterojunction photovoltaic cells with low donor concentration. *Adv. Mater.* 23, 4960–4964 (2011).
- Zou, Y. & Holmes, R. J. The Role of Exciton Ionization Processes in Bulk Heterojunction Organic Photovoltaic Cells. Adv. Energy Mater. 5, 150019 (2015).
- Yan, X. *et al.* The improvement in organic photovoltaic response by inserting an interlayer between MoO3 and mixed layer of C₆₀:5 wt% TAPC. Org. Electron. 23, 5–10 (2015).
- 17. Zou, Y. & Holmes, R. J. Influence of a MoOx interlayer on the open-circuit voltage in organic photovoltaic cells. *Appl. Phys. Lett.* **103**, 053302 (2013).
- Hancox, I. et al. The effect of a MoOx hole-extracting layer on the performance of organic photovoltaic cells based on small molecule planar heterojunctions. Org. Electron. 11, 2019–2025 (2010).
- Irfan et al. Energy level evolution of molybdenum trioxide interlayer between indium tin oxide and organic semiconductor. Appl. Phys. Lett. 96, 073304 (2010).
- Luo, D. Y., Yu, L. M., Man, J. X., Liu, Z. & Lu, Z. H. A nanocomposite interconnecting layer for tandem small molecular organic photovoltaic cells. *Appl. Phys. Lett.* 104, 123301 (2014).
- 21. Cheyns, D., Kim, M., Verreet, B. & Rand, B. P. Accurate spectral response measurements of a complementary absorbing organic tandem cell with fill factor exceeding the subcells. *Appl. Phys. Lett.* **104**, 093302 (2014).
- Shim, H.-S., Chang, J.-H., Yoo, S.-J., Wu, C.-I. & Kim, J.-J. Correlation of the electronic structure of an interconnection unit with the device performance of tandem organic solar cells. J. Mater. Chem. A 2, 5450–5454 (2014).
- Liu, X., Yi, S., Wang, C., Wang, C. & Gao, Y. Electronic structure evolution and energy level alignment at C₆₀/4, 4'-cyclohexylidenebis [N, N-bis (4-methylphenyl) benzenamine]/MoOx/indium tin oxide interfaces. J. Appl. Phys. 115, 163708 (2014).
- 24. Jin, F. et al. Highly efficient organic tandem solar cell based on SubPc: C₇₀ bulk heterojunction. Org. Electron. 15, 3756–3760 (2014).
- Xiao, X., Zimmerman, J. D., Lassiter, B. E., Bergemann, K. J. & Forrest, S. R. A hybrid planar-mixed tetraphenyldibenzoperiflanthene/ C₇₀ photovoltaic cell. App. Phys. Lett. **102**, 073302 (2013).

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Author Contributions

Y.G. wrote the main manuscript and carried out most of the experiments such as OPV device fabrication, measurement, and data analysis. W.L., Z.S. and B.C. provided the necessary consultations during the write-up of the present article. F.J. and J.W. prepared Figures 1–6 and helped measure, collect the experimental data and took part in the mechanism discussions. H.W. and C.L. prepared Figures 7–8 and carried out Tables 1 and 2. T.L. and Q.S. carried out absorption measurements. F.H. and H.Z carried out AFM measurements. All the authors discussed the results and contributed to the article.

Additional Information

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