

Intense Upconversion Luminescence of $\text{CaSc}_2\text{O}_4:\text{Ho}^{3+}/\text{Yb}^{3+}$ from Large Absorption Cross Section and Energy-Transfer Rate of Yb^{3+}

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Concentration-optimized $\text{CaSc}_2\text{O}_4:0.2\%\text{Ho}^{3+}/10\%\text{Yb}^{3+}$ shows stronger upconversion luminescence (UCL) than a typical concentration-optimized upconverting phosphor $\text{Y}_2\text{O}_3:0.2\%\text{Ho}^{3+}/10\%\text{Yb}^{3+}$ upon excitation with a 980 nm laser diode pump. The $^5\text{F}_4+^5\text{S}_2\rightarrow^5\text{I}_8$ green UCL around 545 nm and $^5\text{F}_5\rightarrow^5\text{I}_8$ red UCL around 660 nm of Ho^{3+} are enhanced by factors of 2.6 and 1.6, respectively. On analyzing the emission spectra and decay curves of Yb^{3+} : $^2\text{F}_{5/2}\rightarrow^2\text{F}_{7/2}$ and Ho^{3+} : $^5\text{I}_6\rightarrow^5\text{I}_8$, respectively, in the two hosts, we reveal that Yb^{3+} in CaSc_2O_4 exhibits a larger absorption cross section at 980 nm and subsequent larger Yb^{3+} : $^2\text{F}_{5/2}\rightarrow\text{Ho}^{3+}$: $^5\text{I}_6$ energy-transfer coefficient ($8.55\times 10^{-17}\text{cm}^3\text{s}^{-1}$) compared to that ($4.63\times 10^{-17}\text{cm}^3\text{s}^{-1}$) in Y_2O_3 , indicating that $\text{CaSc}_2\text{O}_4:\text{Ho}^{3+}/\text{Yb}^{3+}$ is an excellent oxide upconverting material for achieving intense UCL.

Owing to its intense green emission under excitation at 455 nm, which is comparable to the commercial yttrium aluminum garnet (YAG): Ce^{3+} phosphor, Ce^{3+} -doped CaSc_2O_4 is considered as one of the most efficient downconversion photoluminescence materials.^[1] Infrared-to-visible upconversion luminescence (UCL) has been extensively studied in various oxide hosts for its potential applications in the detection of infrared light, display technologies, and biological medicine.^[2–5] The CaSc_2O_4 oxide host lattice has a low cutoff phonon frequency of only 540cm^{-1} , which can better inhibit non-radiative multiphonon relaxations in the UCL process, achieving highly efficient luminescence.^[6] Limited work has been carried out on the UCL of rare-earth-metal-doped CaSc_2O_4 materials. Our group has reported the intense UCL in $\text{Tm}^{3+}/\text{Yb}^{3+}$ co-doped CaSc_2O_4 .^[6] The near-infrared emission around 800 nm of Tm^{3+} is enhanced by 3.5 times, compared with that in the typical oxide host Y_2O_3 . Ho^{3+} is one of the most important active ions, owing to its intense green UCL.^[7] But, fewer studies on the UCL of Ho^{3+} -doped materials have been reported in contrast with Tm^{3+} ions. By combining the promising optical properties of Ho^{3+} and good qualities of CaSc_2O_4 , the UCL properties of Ho^{3+} in the CaSc_2O_4 host deserve special attention.

In this Communication, we report a large enhancement of UCL in concentration-optimized $\text{CaSc}_2\text{O}_4:0.2\%\text{Ho}^{3+}/10\%\text{Yb}^{3+}$. UCL intensities around 545 and 660 nm are enhanced by factors of 2.6 and 1.6, respectively, in comparison with that of concentration-optimized $\text{Y}_2\text{O}_3:0.2\%\text{Ho}^{3+}/10\%\text{Yb}^{3+}$. The UCL enhancement is attributed to the larger absorption cross section at 980 nm of Yb^{3+} and subsequent larger $\text{Yb}^{3+}\rightarrow\text{Ho}^{3+}$ energy-transfer coefficient in CaSc_2O_4 than in Y_2O_3 .

A large number experiments show that $\text{CaSc}_2\text{O}_4:\text{Ho}^{3+}/\text{Yb}^{3+}$ exhibits stronger UCL than $\text{Y}_2\text{O}_3:\text{Ho}^{3+}/\text{Yb}^{3+}$. Figure 1 shows

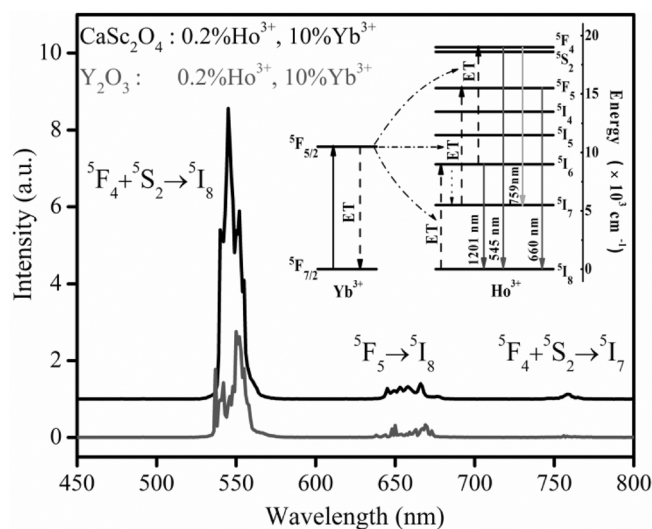


Figure 1. UCL spectra of concentration-optimized $\text{CaSc}_2\text{O}_4:0.2\%\text{Ho}^{3+}/10\%\text{Yb}^{3+}$ and $\text{Y}_2\text{O}_3:0.2\%\text{Ho}^{3+}/10\%\text{Yb}^{3+}$ under 980 nm excitation with a low pump density (7mWmm^{-2}). Inset shows the energy-level diagrams and energy-transfer pathways.

the UCL spectra of concentration-optimized $\text{CaSc}_2\text{O}_4:0.2\%\text{Ho}^{3+}/10\%\text{Yb}^{3+}$ and $\text{Y}_2\text{O}_3:0.2\%\text{Ho}^{3+}/10\%\text{Yb}^{3+}$ under 980 nm excitation with an output power density of 7mWmm^{-2} . The spectra exhibit three UC emissions peaked around 545, 660, and 759 nm, which are assigned to the $^5\text{F}_4+^5\text{S}_2\rightarrow^5\text{I}_8$, $^5\text{F}_5\rightarrow^5\text{I}_8$, and $^5\text{F}_4+^5\text{S}_2\rightarrow^5\text{I}_7$ transitions of Ho^{3+} , respectively.^[10] The pathways for UC emissions are demonstrated schematically under 980 nm excitation in the inset of Figure 1. Compared with $\text{Y}_2\text{O}_3:\text{Ho}^{3+}/\text{Yb}^{3+}$, stronger UCL is observed in $\text{CaSc}_2\text{O}_4:\text{Ho}^{3+}/\text{Yb}^{3+}$. The green (545 nm) and red (660 nm) UCL intensities are enhanced by a remarkable factor of 2.6 and 1.6 in CaSc_2O_4 , respectively.

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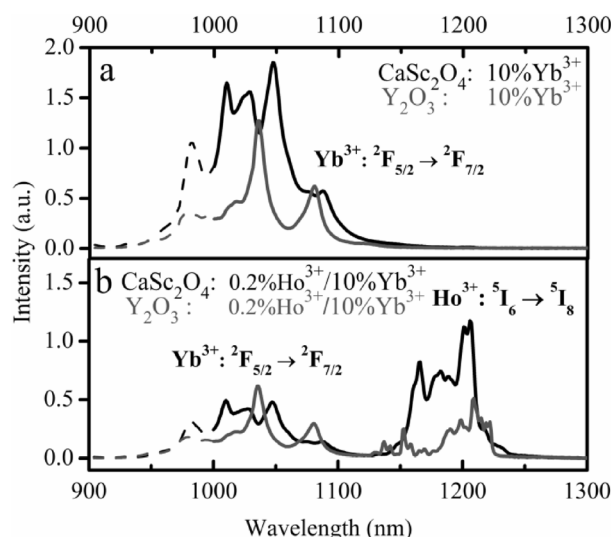


Figure 2. Infrared emission spectra of a) 10% Yb³⁺ singly doped and b) 0.2% Ho³⁺/10% Yb³⁺ co-doped CaSc₂O₄ and Y₂O₃ under 980 nm excitation with the same pump density (7 mW mm⁻²).

To understand the UCL enhancement, we focus on a comparison of the absorption capacities at 980 nm by Yb³⁺ in the two hosts and the energy-transfer rates from Yb³⁺ to Ho³⁺. Figure 2a shows emission spectra of Yb³⁺ singly doped CaSc₂O₄:10%Yb³⁺ and Y₂O₃:10%Yb³⁺ upon 980 nm excitation with an output power density of 7 mW mm⁻². The spectra show typical ²F_{5/2}→²F_{7/2} emissions of Yb³⁺. The area emission intensity (*I*₁) of Yb³⁺ in CaSc₂O₄ is about 2.4 times that in Y₂O₃. As Ho³⁺/Yb³⁺ is co-doped in the two hosts, Ho³⁺:⁵I₆→⁵I₈ emissions around 1200 nm are found, and the area intensity (*I*₂) of which in CaSc₂O₄ is 2.9 times stronger than that in Y₂O₃, as shown in Figure 2b. Meanwhile, the emission of Yb³⁺ in CaSc₂O₄ has a faster reduction. The results indicate that Yb³⁺ absorptivity is larger at a pump wavelength of 980 nm and Yb³⁺→Ho³⁺ energy transfer is more efficient in CaSc₂O₄ than that in Y₂O₃.

Figure 3a shows the decay curves of Yb³⁺:²F_{5/2}→²F_{7/2} emission under pulsed 980 nm excitation in the Yb³⁺ singly doped and Ho³⁺/Yb³⁺ co-doped CaSc₂O₄ and Y₂O₃ samples. As Ho³⁺ is co-doped, the decays rapidly speed up, reflecting remarkable Yb³⁺→Ho³⁺ energy transfer. The Yb³⁺:²F_{5/2} lifetimes (τ_1) for 10% Yb³⁺ singly (τ_{11}) and 0.2% Ho³⁺/10% Yb³⁺ doubly (τ_{12}) doped samples are calculated by integrating the area under the corresponding decay curves with the normalized initial in-

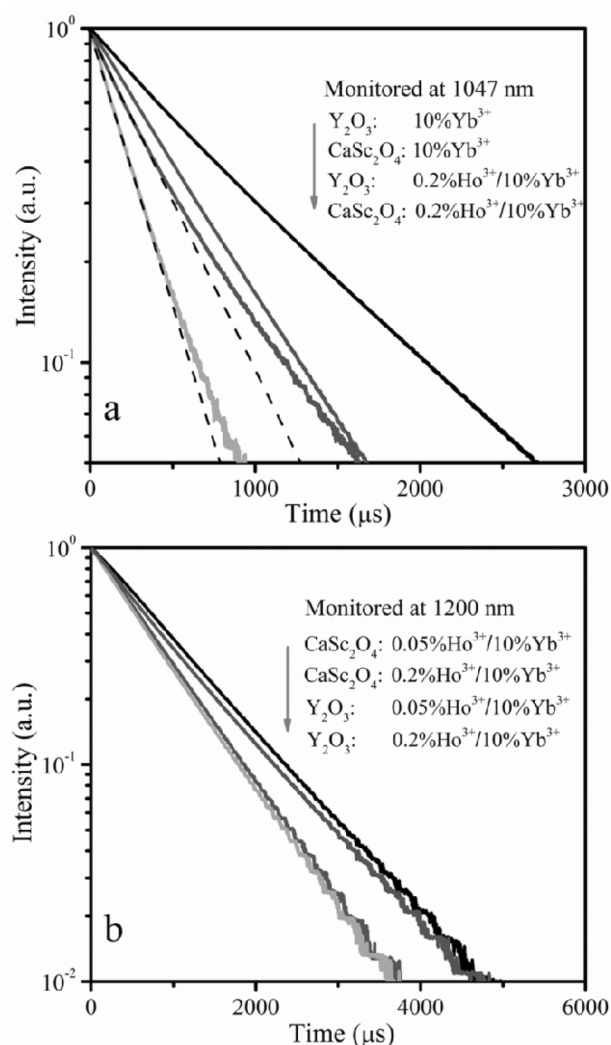


Figure 3. Decay curves of a) Yb³⁺:²F_{5/2}→²F_{7/2} emission at 1047 nm and b) Ho³⁺:⁵I₆→⁵I₈ emission at 1200 nm for doped CaSc₂O₄ and Y₂O₃ under pulsed 980 nm excitation.

tensity, as listed in Table 1. The intrinsic lifetimes (τ_{10}) are taken from our previous studies in dilute Yb³⁺-doped samples.^[6] The emission efficiency (η_1) can be calculated by $\eta_1 = \tau_1/\tau_{10}$. The Yb³⁺→Ho³⁺ energy-transfer efficiency (η_{ET}) can be calculated by $\eta_{ET} = 1 - \tau_{12}/\tau_{11}$. Hence, the quenching efficiency (η_q) is deduced by $1 - \eta_1 - \eta_{ET}$.

Figure 3b shows the decay curves of the Ho³⁺:⁵I₆→⁵I₈ emission for the 0.05% Ho³⁺ or 0.2% Ho³⁺ with 10% Yb³⁺ doubly

Sample	<i>I</i> ₁	τ_1 [μ s]	η_1	η_q	η_{ET}	<i>A</i>	σ	<i>C</i> [cm ³ s ⁻¹]	<i>x_C</i> [cm ⁻³]
CaSc ₂ O ₄ :0.2% Yb ³⁺ (4)	2.4 θ_1	τ_{10} : 671	1	0	0	2.7	2.9	8.55 × 10 ⁻¹⁷	1.74 × 10 ¹⁹
CaSc ₂ O ₄ :10% Yb ³⁺		τ_{11} : 548	0.81	0.19	0				
CaSc ₂ O ₄ :0.2% Ho ³⁺ /10% Yb ³⁺		τ_{12} : 286	0.43	0.09	0.48				
Y ₂ O ₃ :0.2% Yb ³⁺ (4)	θ_1	τ_{10} : 932	1	0	0	1	1	4.63 × 10 ⁻¹⁷	2.32 × 10 ¹⁹
Y ₂ O ₃ :10% Yb ³⁺		τ_{11} : 835	0.90	0.10	0				
Y ₂ O ₃ :0.2% Ho ³⁺ /10% Yb ³⁺		τ_{12} : 479	0.51	0.07	0.42				

doped CaSc_2O_4 and Y_2O_3 samples under pulsed 980 nm excitation. $\text{Ho}^{3+}:^5\text{I}_6$ exhibits a single exponential decay, indicating no back energy-transfer from Ho^{3+} to Yb^{3+} . The decay for CaSc_2O_4 is much slower than that observed in the Y_2O_3 host. The long-lived intermediate state, $^5\text{I}_6$, is the basis of intense green UCL.^[10] The average lifetimes (τ_2) and emission efficiencies (η_2) of the $\text{Ho}^{3+}:^5\text{I}_6$ level obtained from these decay patterns are listed in Table 2. It is found that the τ_2 gets slightly shorter when increasing the Ho^{3+} concentration from 0.05 to 0.2%, indicating a weak quenching. The η_2 value is close to 1.

Table 2. Intensities and lifetimes of Ho^{3+} emissions in CaSc_2O_4 and Y_2O_3 .

Sample	I_2	τ_2 [μs]	η_2	I_3	τ_3 [μs]	I_4	τ_4 [μs]
$\text{CaSc}_2\text{O}_4:0.05\% \text{Ho}^{3+}, 10\% \text{Yb}^{3+}$		1039	1				
$\text{CaSc}_2\text{O}_4:0.2\% \text{Ho}^{3+}, 10\% \text{Yb}^{3+}$	$2.9\theta_2$	960	0.92	$1.6\theta_3$	410	$2.6\theta_4$	249
$\text{Y}_2\text{O}_3:0.05\% \text{Ho}^{3+}, 10\% \text{Yb}^{3+}$		812	1				
$\text{Y}_2\text{O}_3:0.2\% \text{Ho}^{3+}, 10\% \text{Yb}^{3+}$	θ_2	774	0.95	θ_3	395	θ_4	224

For Yb^{3+} singly doped samples, the absorptivity (A) of Yb^{3+} at 980 nm can be estimated by $I_1 = PA\eta_1$, where P is the pump density. The absorptivity of Yb^{3+} in $\text{CaSc}_2\text{O}_4:10\% \text{Yb}^{3+}$ is 2.7 times bigger than that in $\text{Y}_2\text{O}_3:10\% \text{Yb}^{3+}$. When Ho^{3+} is co-doped, the efficient $\text{Yb}^{3+} \rightarrow \text{Ho}^{3+}$ energy transfer is observed. Owing to the fact that Ho^{3+} ions are almost in the ground state under weak excitation in this work, the calculated transfer efficiency (η_{ET}) is dominantly responsible for the energy transfer to excite Ho^{3+} from its ground state, $^5\text{I}_8$, to $^5\text{I}_6$. So, the absorptivity of Yb^{3+} in the $\text{Ho}^{3+}/\text{Yb}^{3+}$ doubly doped samples can be estimated by using η_{ET} , the measured emission intensity (I_2), and the efficiency (η_2) of $\text{Ho}^{3+}:^5\text{I}_6$, which satisfy the relationship $I_2 = PA\eta_{\text{ET}}\eta_2$. The ratio of absorptivity in CaSc_2O_4 to that in Y_2O_3 is calculated to be 2.6 for the $0.2\% \text{Ho}^{3+}/10\% \text{Yb}^{3+}$ doubly doped samples. The two values of A ratios are consistent before and after energy transfer, showing the validity of the obtained A values. The ratio of absorption cross section (σ) of Yb^{3+} at 980 nm in CaSc_2O_4 to that in Y_2O_3 is calculated to be 2.9 for singly doped $10\% \text{Yb}^{3+}$ and 2.8 for $0.2\% \text{Ho}^{3+}/10\% \text{Yb}^{3+}$ doubly doped samples, using the value of A/N , where N is the Yb^{3+} number per volume, $1.2 \times 10^{21} \text{ cm}^{-3}$ in CaSc_2O_4 and $1.3 \times 10^{21} \text{ cm}^{-3}$ in Y_2O_3 .

It can be also seen in Figure 3a that the initial decay rate of $\text{Yb}^{3+} \rightarrow \text{Ho}^{3+}$ energy transfer lasts longer in CaSc_2O_4 than Y_2O_3 , which is beneficial for highly effective energy transfer. The initial energy transfer rate, W_{ET} , could be calculated by $W_{12i} - 1/\tau_{11}$, where W_{12i} is the initial decay rate for the doubly doped samples. The $\text{Yb}^{3+} \rightarrow \text{Ho}^{3+}$ energy-transfer coefficient (C) at $t = 0$ can be expressed as $W_{\text{ET}} = Cx$, where x is the Ho^{3+} concentration, which was $2.4 \times 10^{19} \text{ cm}^{-3}$ in CaSc_2O_4 and $2.6 \times 10^{19} \text{ cm}^{-3}$ in Y_2O_3 , in this work.^[11] Then, we obtained a transfer coefficient of $8.55 \times 10^{-17} \text{ cm}^3 \text{ s}^{-1}$ for $\text{CaSc}_2\text{O}_4:0.2\% \text{Ho}^{3+}/10\% \text{Yb}^{3+}$ and $4.63 \times 10^{-17} \text{ cm}^3 \text{ s}^{-1}$ for $\text{Y}_2\text{O}_3:0.2\% \text{Ho}^{3+}/10\% \text{Yb}^{3+}$. The former transfer coefficient was twice as large as the latter. The corresponding critical Ho^{3+} concentration x_c values, defined as $Cx_c = 1/\tau_{10}$, were determined to be 1.74×10^{19} and $2.32 \times 10^{19} \text{ cm}^{-3}$, respectively.

The lifetimes associated with the red (τ_3) and green (τ_4) emissions in $0.2\% \text{Ho}^{3+}/10\% \text{Yb}^{3+}$ co-doped samples are also listed in Table 2 under pulsed 980 nm excitation. The values of τ_3 and τ_4 are both longer in CaSc_2O_4 than in Y_2O_3 . The lifetime of an energy level is proportional to the population.^[10] It also indicates stronger red and green UCL in $\text{CaSc}_2\text{O}_4:0.2\% \text{Ho}^{3+}/10\% \text{Yb}^{3+}$.

The CaSc_2O_4 lattice, which owns an orthorhombic CaFe_2O_4 structure with the space group $Pnam$ (62) exhibits a lower crystal-field symmetry than cubic Y_2O_3 . Low-symmetry hosts exert a crystal field containing more uneven components around the dopant ions, which can improve the transition probabilities.^[12] The $\text{Yb}^{3+} \rightarrow \text{Ho}^{3+}$ energy-transfer efficiency in $\text{CaSc}_2\text{O}_4:0.2\% \text{Ho}^{3+}/10\% \text{Yb}^{3+}$ is larger than that in $\text{Y}_2\text{O}_3:0.2\% \text{Ho}^{3+}/10\% \text{Yb}^{3+}$, as shown in Table 1. This can be attributed to the fact that the ground-state splitting of Yb^{3+} (1008 cm^{-1}) in CaSc_2O_4 is larger than that (931 cm^{-1}) in Y_2O_3 , owing to the fact that Yb^{3+} on the Sc^{3+} site in CaSc_2O_4 experiences a stronger crystal field than Yb^{3+} on the Y^{3+} site in Y_2O_3 , and the mean distance between the nearest two Sc atoms (Sc–Sc) in CaSc_2O_4 is 3.186 \AA , which is much shorter than the Y–Y distance (3.752 \AA) in Y_2O_3 .^[6] Considering the equipment exists, the response time to the 980 nm laser, which is around $73 \mu\text{s}$ (according to our lifetime measurement), and fast initial decays caused by energy transfer of the nearest neighbor ions cannot be observed ($\approx \text{ns}$).^[3,13] The real lifetime value was lower than the experimental one in our study. Reabsorption effects also cause significant lengthening of the measured fluorescence lifetimes.^[14] The real η_{ET} values were larger than the experimental data. But, the relative ratio of η_{ET} in the two hosts was less affected by it, which could be used for our above physical analysis.^[14]

Compared with energy-transfer efficiency, the larger absorption cross section at 980 nm of Yb^{3+} (three times that in Y_2O_3) plays an important role in the achievement of the strong $\text{Ho}^{3+}:^5\text{I}_6 \rightarrow ^5\text{I}_8$ emission in the CaSc_2O_4 sample. As a result, the 2.6-fold enhanced green and 1.6-fold enhanced red emissions were observed in $\text{CaSc}_2\text{O}_4:0.2\% \text{Ho}^{3+}/10\% \text{Yb}^{3+}$. The green emission inherits the enhanced emission from $\text{Ho}^{3+}:^5\text{I}_6$. The acquired enlarged values of intensities are comparable. The smaller increase in the red UCL is perhaps a result of the weak multiphonon relaxation from $^5\text{I}_6$ to $^5\text{I}_7$ in CaSc_2O_4 . The energy gap between the $^5\text{I}_6$ and $^5\text{I}_7$ levels is around 3240 cm^{-1} .^[15] The CaSc_2O_4 oxide host has a lower cutoff phonon frequency of 540 cm^{-1} compared to Y_2O_3 (600 cm^{-1}), which suppresses non-radiative relaxation. This explanation requires evidence, which will be obtained by further study in the future.

In summary, we observed a largely enhanced UCL in $\text{CaSc}_2\text{O}_4:0.2\% \text{Ho}^{3+}/10\% \text{Yb}^{3+}$ upon excitation with a 980 nm laser diode pump. Compared with concentration-optimized $\text{Y}_2\text{O}_3:0.2\% \text{Ho}^{3+}/10\% \text{Yb}^{3+}$, the green and red UCL were enhanced by a factor of 2.6 and 1.6, respectively. The larger absorption cross section at 980 nm for Yb^{3+} (three times that of

Y_2O_3) and the $\text{Yb}^{3+}:^2\text{F}_{5/2} \rightarrow \text{Ho}^{3+}:^5\text{I}_6$ energy-transfer coefficient (twice that of Y_2O_3) play important roles for the achievement of intense UCL in the CaSc_2O_4 phosphor. $\text{CaSc}_2\text{O}_4:\text{Ho}^{3+}/\text{Yb}^{3+}$ is a promising upconverting oxide material for achieving highly efficient green UCL with diverse applications.

Experimental Section

The doped CaSc_2O_4 powder samples were synthesized by using a common solid-state reaction at 1500°C for 4 h.^[8] Y_2O_3 samples for comparison with CaSc_2O_4 were prepared through the reported sol-gel method.^[9] Our experiments showed that the sol-gel method for the preparation of Y_2O_3 is more favorable for the achievement of high crystallinity, uniform distribution of active ions (Yb^{3+} and Ho^{3+}), and intense UCL than the solid-state reaction. Both of the produced CaSc_2O_4 and Y_2O_3 samples had a high crystallinity. The $\text{CaSc}_2\text{O}_4:0.2\%\text{Ho}^{3+}/10\%\text{Yb}^{3+}$ and $\text{Y}_2\text{O}_3:0.2\%\text{Ho}^{3+}/10\%\text{Yb}^{3+}$ samples were optimized for the highest green UCL intensity in each host.^[9,10] The Ho^{3+} and Yb^{3+} ions occupied Sc^{3+} sites in CaSc_2O_4 or Y^{3+} sites in Y_2O_3 . The UCL spectra were measured by using a Triax 550 spectrometer (Jobin-Yvon) pumped with a power-controllable 980 nm diode laser. In fluorescence-lifetime measurements, a pulsed 980 nm laser of an optical parametric oscillator (OPO) was used as an excitation source, and the signals were detected by a Tektronix digital oscilloscope (TDS 3052).

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