

THE CHARACTERISTIC OF ACEL IN ZnS:Er^{3+} THIN FILMS GROWN BY ATOMIC LAYER EPITAXY METHOD(ALE)

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The characteristics of ACEL in ZnS:Er^{3+} thin films grown by ALE method were reported for the first time. The cross-relaxation model between Er^{3+} ions was set up. The cross-relaxation rate was calculated.

1. INTRODUCTION

The work of electroluminescence in $\text{A}^{\text{II}}\text{B}^{\text{VI}}$ thin films doped with rare earth ions initiated in 1968 by Kahng^[1]. Recently, H. Kobayashi et al^[2], have reported ACEL characteristics of ZnS thin films doped with TbF_3 , SmF_3 and TmF_3 , respectively. They have studied the interaction between ions and given cross-relaxation model.

In this paper we are reporting the characteristics of ACEL in ZnS:Er^{3+} thin films grown by ALE method^[3] and discussing the cross-relaxation processes between Er^{3+} ions. The results show that the cross-relaxation processes will occur between excited Er^{3+} ions.

2. EXPERIMENTAL

The EL devices are glass- SnO_2 - Y_2O_3 - ZnS:Er^{3+} - Y_2O_3 -Al sandwich structure. The Y_2O_3 insulator layer was fabricated by electro-beam evaporation ZnS:Er^{3+} layer was grown by ALE method. The ALE is based on chemical reaction in the solid surface of a substrate to which the reactants are alternately transported as neutral molecular or atomic vapor pulses (or beams)^[4]. The thicknesses of Y_2O_3 layer and ZnS:Er^{3+} active layer were about 300nm and 400nm, respectively. The concentration of Er^{3+} ions in ZnS thin films ranges from 0.05% to 12%(wt).

3. RESULTS AND DISCUSSION

Two typical samples with different Er^{3+} concentration were investigated. The Er^{3+} concen-

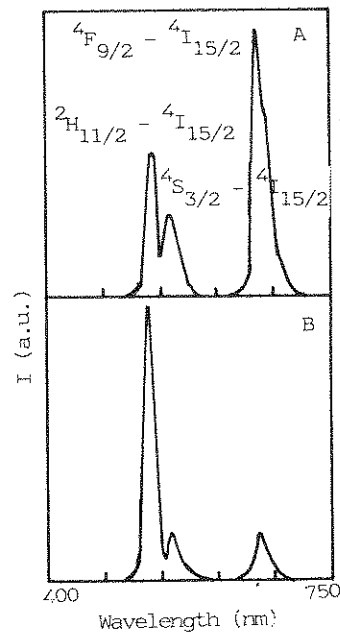


FIGURE 1
ACEL spectra of samples A and B at RT.

tration in samples A and B are 12% and 0.9%(wt), respectively. The EL spectra of both samples at room temperature(RT) showed that the emission intensity of $4\text{F}_{9/2}-4\text{I}_{15/2}$ transition increased strongly with increasing of Er^{3+} concentration. $I(4\text{F}_{9/2}-4\text{I}_{15/2}) > I(2\text{H}_{11/2}-4\text{I}_{15/2})$ for sample A and $I(4\text{F}_{9/2}-4\text{I}_{15/2}) < I(2\text{H}_{11/2}-4\text{I}_{15/2})$ for sample B, as shown in Fig. 1.

EL decay curves of sample A are different from that of sample B. The decay curves of

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26

E (X 10³ / cm)

16

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relaxation rate

A
5/2
B
750

²H_{11/2}, ⁴S_{3/2} and ⁴F_{9/2} state in sample B are single exponential law. However, for sample A, there is a rising process in the EL decay of ⁴F_{9/2} state and there are a fast and a slow component in the EL decay of ²H_{11/2} and ⁴S_{3/2} state. The EL decay curves of ²H_{11/2} and ⁴S_{3/2} state can approximately be expressed as

$$J = J_{10} * \text{EXP}(-t/t_1) + J_{20} * \text{EXP}(-t/t_2)$$

where J₁₀ and J₂₀ are initial brightnesses of slow and fast component, t₁ and t₂ are lifetimes of slow and fast component, respectively.

From above results, we presented an interpreta-

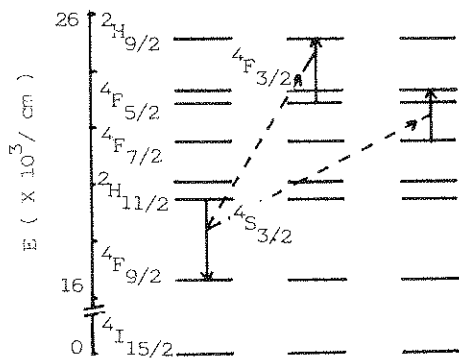


FIGURE 2
Possible cross-relaxation processes occurred between Er³⁺ ions.

tation; the fast component of EL decay curves of ²H_{11/2} and ⁴S_{3/2} state results from relaxation from ⁴S_{3/2} and ²H_{11/2} state to ⁴F_{9/2} state. These processes will enhance the emission intensity of ⁴F_{9/2}-⁴I_{15/2} transition. Nonradiative relaxation between electronic state of Er³⁺ ions occurs usually by multi-phonon emission or by cross-relaxation. The former has been expected^[5]. By analysis of experimental results we may suppose the concrete cross-relaxation processes, as shown in Fig. 2.

The cross-relaxation rate of sample A calculated by the experimental data was about 1.4*10⁵ per second. This value was in agreement with that value(1.3*10⁵ per second) obtained by simulating decay curve.

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