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Sensitized luminescence and energy transfer in Ce³⁺ and Eu²⁺ codoped calcium magnesium chlorosilicate

H. Lin a,b, X.R. Liu b, E.Y.B. Pun a,*

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Abstract

 Ce^{3+} and Eu^{2+} ions codoped calcium magnesium chlorosilicate $[Ca_8Mg(SiO_4)_4Cl_2]$ phosphors have been synthesized and characterized. Intense bluish-green light was observed under ultraviolet and blue light excitations. The diffuse reflection, excitation and emission spectra of $Ca_8Mg(SiO_4)_4Cl_2:Ce^{3+}$, Eu^{2+} were measured at room temperature, and enhancement of Eu^{2+} emission was confirmed. The fluorescence lifetimes of Ce^{3+} were measured, and the mechanism of nonradiative energy transfer from Ce^{3+} to Eu^{2+} is found to be electric dipole–dipole interaction. The probabilities and efficiencies of nonradiative energy transfer were also calculated, and the results indicate that the efficiency can be as high as 86% in this material system. © 2002 Published by Elsevier Science B.V.

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1. Introduction

Solid-state green light sources are desirable for high density optical storage, color displays, medical diagnostics and undersea communications [1,2]. Many efforts have been spent on ${\rm Er}^{3+}$ doped fluoride materials due to their efficient green upconversion [1–3]. However, these materials are not stable and may be not suitable for device applications. Calcium magnesium chlorosilicate $[{\rm Ca_8Mg}({\rm SiO_4})_4{\rm Cl_2}]$ is a new kind of luminescence

E-mail address: eeeybpun@cityu.edu.hk (E.Y.B. Pun).

host with stable crystal structure and high thermal stability, and Eu^{2+} and Ce^{3+} singly doped $Ca_8Mg(SiO_4)_4Cl_2$ have been demonstrated to be high efficient green and purplish-blue phosphors, respectively [4,5]. $Ca_8Mg(SiO_4)_4Cl_2:Eu^{2+}$ phosphor emits bright green light when excited by ultraviolet and blue lights from near ultraviolet- and blue-emitting LEDs, and can be used to make efficient green light sources. In addition, this stable phosphor is important for the development of materials against counterfeiting [6] and efficient ultraviolet-visible transformation devices.

It is well known that the energy transfer from donor to acceptor plays an important role in luminescence and solid-state lasers [7,8]. The 4f-5d

^a Department of Electronic Engineering, City University of Hong Kong, Tat Chee Avenue, Kowloon, Hong Kong, People's Republic of China

b Changchun Institute of Optics, Fine Mechanics and Physics, Chinese Academy of Sciences, Changchun 130022, People's Republic of China

^{*}Corresponding author. Tel.: 85-2-2788-8609; fax: +85-2-2788-7791.

transitions of Ce^{3+} and Eu^{2+} are very sensitive to the nature of the host lattice, and the absorption and emission of Ce^{3+} and Eu^{2+} are efficient in many hosts [9–12]. In Ce^{3+} and Eu^{2+} codoped $Ca_8Mg(SiO_4)_4Cl_2$ new system, strong coupling between Ce^{3+} and Eu^{2+} ions may occur, leading to variations in the excitation and emission spectra of Eu^{2+} . This is because the Ce^{3+} emission spectrum in the purplish-blue region overlaps considerably with the excitation spectrum of Eu^{2+} . Moreover, for practical applications it would be important if Ce^{3+} could greatly enhance the emission of Eu^{2+} , because cheaper CeO_2 could be used partly to replace expensive Eu_2O_3 .

In this paper, the luminescence properties of and the energy transfer process in Ca₈Mg(SiO₄)₄ Cl₂:Ce³⁺, Eu²⁺ are reported. The diffuse reflection, excitation and emission spectra, and the fluorescence intensities of Ca₈Mg(SiO₄)₄Cl₂:Ce³⁺, Eu²⁺ are compared to those of Eu²⁺ singly doped Ca₈Mg(SiO₄)₄Cl₂. The enhanced green emission of Eu²⁺ is confirmed, and there is an efficient nonradiative energy transfer between Ce³⁺ and Eu²⁺ under ultraviolet light excitation. Fluorescence decay curves of Ce³⁺ in Ca₈Mg(SiO₄)₄Cl₂:Ce³⁺, Eu²⁺ system were obtained, and the experimental decay curves were fitted to the equations given by Inokuti and Hirayama (IH model) [13].

2. Experimental

Polycrystalline samples of $Ca_8Mg(SiO_4)_4$ $Cl_2:Ce^{3+}$, Eu^{2+} were synthesized by high temperature solid state reaction. The starting materials were SiO_2 (GR), $CaCO_3$ (AR), MgO (4N), $CaCl_2$ (AR), Eu_2O_3 (4N), CeO_2 (4N), and Li_2CO_3 (AR). The synthesis procedure is similar to that described previously [4,5].

The crystal structure of the powder samples was examined using a D/max-rA X-ray diffractometer. The diffraction patterns indicate that the materials obtained were cubic calcium magnesium chlorosilicate. The excitation and fluorescence spectra of Ca₈Mg(SiO₄)₄Cl₂:Ce³⁺, Eu²⁺ were measured at room temperature using a Hitachi MPF-4 spectrofluorometer and a 75-W xenon lamp source. The diffuse reflection spectra of the samples were

measured relative to MgO, and the wavelength resolution for all spectra was 0.5 nm.

The decay curves of the 5d level of Ce³⁺ ions in the Ca₈Mg(SiO₄)₄Cl₂:Ce³⁺, Eu²⁺ system were measured using time-correlated single-photon counting technique. The excitation source was a nanosecond flash lamp filled with nitrogen, and the excitation wavelength was 337 nm. The distortion of the fluorescence due to excitation source, photomultiplier and electronic system was removed by a suitable deconvolution procedure.

3. Results and discussion

The polycrystalline $Ca_8Mg(SiO_4)_4Cl_2$ samples were white in colour. $Ca_8Mg(SiO_4)_4Cl_2$: Eu^{2+} and $Ca_8Mg(SiO_4)_4Cl_2$: Eu^{2+} phosphors were yellowish-green in colour. The diffuse reflection spectra of $Ca_8Mg(SiO_4)_4Cl_2$ host, $Ca_8Mg(SiO_4)_4$ Cl_2 : $0.006Ce^{3+}$, $Ca_8Mg(SiO_4)_4Cl_2$: $0.03Eu^{2+}$ and $Ca_8Mg(SiO_4)_4Cl_2$: $0.006Ce^{3+}$, $0.03Eu^{2+}$ phosphors are shown in Fig. 1. Curve 1 indicates that the absorption of $Ca_8Mg(SiO_4)_4Cl_2$ host is strong in the 240–340 nm spectral region, weak in the ultraviolet long-wavelength region, and most of excitation energy is reflected in the visible light region. Comparison of curves 2 and 3 with 1 indicates that the absorption of $Ca_8Mg(SiO_4)_4$ Cl_2 : $0.006Ce^{3+}$ and $Ca_8Mg(SiO_4)_4Cl_2$: $0.03Eu^{2+}$ in

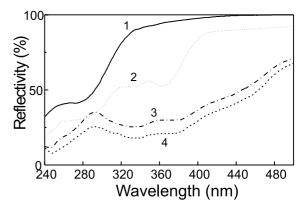


Fig. 1. Diffuse reflection spectra of $Ca_8Mg(SiO_4)_4Cl_2$ host (curve 1), $Ca_8Mg(SiO_4)_4Cl_2:0.006Ce^{3+}$ (curve 2), $Ca_8Mg(SiO_4)_4Cl_2:0.03Eu^{2+}$ (curve 3) and $Ca_8Mg(SiO_4)_4Cl_2:0.006Ce^{3+},0.03Eu^{2+}$ (curve 4).

the 320-480 nm spectral range is much stronger than that of the Ca₈Mg(SiO₄)₄Cl₂ host. The two diffuse reflection peaks of Ca₈Mg(SiO₄)₄Cl₂:0.006 Ce^{3+} are located at ~333 and 363 nm, and correspond to the 4f¹-4f⁰5d¹ transitions of Ce³⁺ in Ca₈Mg(SiO₄)₄Cl₂. The diffuse reflection of Ca₈ Mg(SiO₄)₄Cl₂:0.03Eu²⁺ consists of three broad bands, and the peaks are at 334, 378 and \sim 450 nm. These absorption bands belong to the 4f⁷-4f⁶5d¹ transitions of Eu²⁺ and are strong in the ultraviolet and blue spectral regions. This is an essential condition for Ca₈Mg(SiO₄)₄Cl₂:Eu²⁺ to become an efficient green luminescence material. The diffuse reflection of Ca₈Mg(SiO₄)₄Cl₂:Ce³⁺, Eu²⁺ (curve 4) shows that the absorption in the 240-500 nm spectral range is stronger than those of $Ca_8Mg(SiO_4)_4Cl_2$ and $Ca_8Mg(SiO_4)_4Cl_2:Eu^{2+}$, because the absorption is the net result of Ca₈Mg $(SiO_4)_4Cl_2$ host and the 4f \rightarrow 5d band transition absorption of Eu²⁺ and Ce³⁺. This efficient absorption of excitation energy increases the luminescence efficiency of phosphors.

The emission spectra of $Ca_8Mg(SiO_4)_4Cl_2$: XCe^{3+} , $0.03Eu^{2+}$ and $Ca_8Mg(SiO_4)_4Cl_2$: $0.006Ce^{3+}$ at room temperature under 365 nm ultraviolet light excitation are shown in Fig. 2. It can be seen that $Ca_8Mg(SiO_4)_4Cl_2$: Eu^{2+} (curve 1) has a strong green band with the peak at 509 nm which is assigned to the octa-coordinated Eu^{2+} emission center [4]. The emission spectra of $Ca_8Mg(SiO_4)_4$

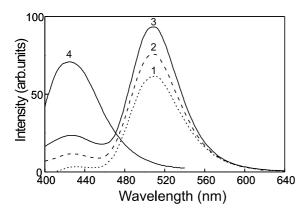


Fig. 2. Fluorescence spectra of $Ca_8Mg(SiO_4)_4Cl_2:0.03Eu^{2+}$ (curve 1), $Ca_8Mg(SiO_4)_4Cl_2:0.006Ce^{3+}$, $0.03Eu^{2+}$ (curve 2), $Ca_8Mg(SiO_4)_4Cl_2:0.03Ce^{3+}$, $0.03Eu^{2+}$ (curve 3) and $Ca_8Mg(SiO_4)_4$ $Cl_3:0.006Ce^{3+}$ (curve 4) under 365 nm excitation.

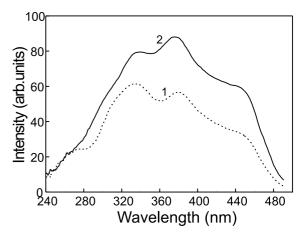


Fig. 3. Excitation spectra of Eu^{2+} (509 nm emission) in $Ca_8Mg(SiO_4)_4Cl_2{:}0.03Eu^{2+}$ (curve 1) and $Ca_8Mg(SiO_4)_4Cl_2{:}0.03Ce^{3+},0.03Eu^{2+}$ (curve 2).

 $\text{Cl}_2:\text{Ce}^{3+}$, Eu^{2+} (curve 2,3) exhibit a weak purplishblue band at 427 nm which is due to the 5d–4f transition of Ce^{3+} , and a strong Eu^{2+} green emission band at 509 nm which is due to the simultaneous excitation of Eu^{2+} . Comparison of curves 1, 2 and 3 shows that codoping with a small amount of Ce^{3+} ions not only enhances the Eu^{2+} green emission band but also changes the luminescence colour of $\text{Ca}_8\text{Mg}(\text{SiO}_4)_4\text{Cl}_2:\text{Eu}^{2+}$ from green to bluish-green.

The excitation spectra of Eu²⁺ (509 nm emission) in both Ca₈Mg(SiO₄)₄Cl₂:Eu²⁺ and Ca₈Mg (SiO₄)₄Cl₂:Ce³⁺, Eu²⁺ are shown in Fig. 3. The Eu²⁺ emission intensity in Ca₈Mg(SiO₄)₄Cl₂: Ce³⁺, Eu²⁺ (curve 2) is higher than that in Ca₈Mg(SiO₄)₄Cl₂:Eu²⁺ (curve 1) under ultraviolet and blue light excitation from 270 to 480 nm spectral range. There are some differences between the spectrum shapes of the two phosphors. In order to show the differences clearly, the spectrum [*I*(curve 2)–*I*(curve 1)] is depicted as curve 1 in Fig. 4.

Curves 1 and 2 [excitation spectrum of $Ca_8Mg(SiO_4)_4Cl_2:Ce^{3+}$] in Fig. 4 are similar in the 270–380 nm spectral region. This means that variations of Eu^{2+} emission intensities are due to the Ce^{3+} excitation band under ultraviolet radiation from 270 to 380 nm, and there is a nonradiative energy transfer process from Ce^{3+} to Eu^{2+} in

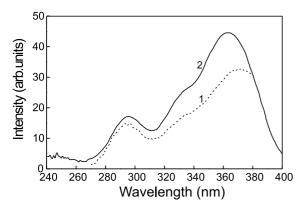


Fig. 4. Curve 1: Difference in excitation spectrum between $Ca_8Mg(SiO_4)_4Cl_2:0.03Ce^{3+},0.03Eu^{2+}$ and $Ca_8Mg(SiO_4)_4Cl_2:0.03Eu^{2+}$. Curve 2: Excitation spectrum of Ce^{3+} (425 nm emission) in $Ca_8Mg(SiO_4)_4Cl_2:0.02Ce^{3+}$.

Ca₈Mg(SiO₄)₄Cl₂:Ce³⁺, Eu²⁺ under ultraviolet light excitation.

Fluorescence decay curves for the $4f^05d^1 \rightarrow 4f^15d^0$ transition of Ce^{3+} in $Ca_8Mg(SiO_4)_4Cl_2$: $0.02Ce^{3+}$, YEu^{2+} samples with different Eu^{2+} concentrations (Y) are shown in Fig. 5. The decay curves are nonexponential with increasing Y. At low donor (Ce^{3+}) concentration, diffusion of excitation energy between Ce^{3+} ions can be neglected. Therefore, the Inokuti–Hiroyama formula [13]

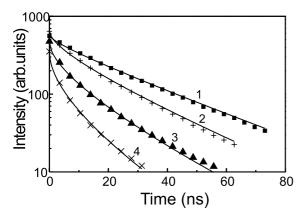


Fig. 5. Fluorescence decay curves of Ce^{3+} (400 nm emission) in $Ca_8Mg(SiO_4)_4Cl_2:0.02Ce^{3+}$, YEu^{2+} . Y=0.02 (no. 1), 0.04 (no. 2), 0.06 (no. 3), and 0.08 (no. 4). Solid curves: IH theoretical curves; symbols: Experimental data.

$$I(t) = I(0) \exp\left[-\frac{t}{\tau_0} - \frac{4}{3}\pi\Gamma(1 - \frac{3}{s})N_A R_0^3 \left(\frac{t}{\tau_0}\right)^{3/s}\right]$$
(1)

can be used to describe the density of the donor (Ce^{3+}) in the presence of the acceptor (Eu^{2+}) , where τ_0 is the intrinstic lifetime of donor ion, N_A is the concentration of acceptor ions, Γ is the gamma function, R_0 is the critical transfer distance, and the interaction parameter s=6 for electric dipole–dipole, s=8 for dipole–quadrupole, and s=10 for quadrupole–quadrupole interaction.

The experimental decay curves were fitted to Eq. (1) by the standard least-squares method. All the root-mean-square (rms) errors are listed in Table 1. The fittings were good in the case of electric dipole–dipole interaction (solid lines in Fig. 5), but became distinctly poorer for the dipole–quadrupole and quadrupole–quadrupole interactions for all these curves. In the last two cases, the rms errors were systematically larger than that for the dipole–dipole interaction by at least a factor of two. The results indicate that dipole–dipole interaction is the dominant mechanism governing the nonradiative energy transfer from Ce^{3+} to Eu^{2+} in $Ca_8Mg(SiO_4)_4Cl_2$: Ce^{3+} , Eu^{2+} under ultraviolet radiation.

The critical energy-transfer distance (R_0) is an adjustable parameter and has an average value of 1.5 nm. The values of the measured lifetime and the derived values of the critical distance are given in Table 2. The probability (P) and the efficiency (η) of nonradiative energy transfer were calculated from the lifetime data using the expressions

$$P = 1/\tau - 1/\tau_0,\tag{2}$$

$$\eta = \frac{P}{1/\tau} = 1 - \frac{\tau}{\tau_0},\tag{3}$$

where τ_0 is the intrinsic lifetime of the donor and τ is the lifetime of the donor in the presence of acceptor. The values of P and η obtained at room temperature are also presented in Table 2, and the efficiency of nonradiative energy transfer can be as high as 86%.

Table 1
Root-mean-square error of fittings on experimental decay curves using Inokuti-Hiroyama formula

Sample	S = 6 (%)	S = 8 (%)	S = 10 (%)
$Ca_8Mg(SiO_4)_4Cl_2:0.02Ce^{3+}, 0.02Eu^{2+}$	2.82	8.41	6.63
$Ca_8Mg(SiO_4)_4Cl_2:0.02Ce^{3+}, 0.04Eu^{2+}$	2.51	6.84	9.73
$Ca_8Mg(SiO_4)_4Cl_2:0.02Ce^{3+}, 0.06Eu^{2+}$	2.42	6.85	10.24
$Ca_8Mg(SiO_4)_4Cl_2:0.02Ce^{3+},0.08Eu^{2+}$	1.69	5.86	8.83

Table 2 Fluorescence lifetimes and energy-transfer parameters of Ce^{3+} in $Ca_8Mg(SiO_4)_4Cl_2$: XCe^{3+} , YEu^{2+}

Ce^{3+} conc. (X)	Eu ²⁺ conc. (Y)	τ (ns)	rms (%)	η (%)	$P \ (\times 10^7 \ \mathrm{s}^{-1})$	R_0 (nm)
0.02	0	35.47				
0.02	0.02	24.10	4.14	32	1.33	1.41
0.02	0.04	14.96	10.41	58	3.87	1.48
0.02	0.06	8.85	15.11	75	8.48	1.52
0.02	0.08	5.01	13.69	86	17.14	1.58

4. Conclusion

In summary, Ca₈Mg(SiO₄)₄Cl₂:Ce³⁺, Eu²⁺ has been demonstrated to be an efficient bluish-green emission phosphor under ultraviolet light excitation. In this material system, Ce³⁺ and Eu²⁺ act as donor and acceptor, respectively, and the efficiency of nonradiative energy transfer from Ce³⁺ to Eu²⁺ can be as high as 86%. Comparison of the excitation and emission spectra indicates that Ce³⁺ is a sensitizer in this system. The florescence decay curves of Ce³⁺ in this material system were fitted to the Inokuti-Hirayama formula, and the dominant nonradiative energy transfer mechanism from Ce^{3+} to Eu^{2+} is found to be electric dipole–dipole interaction. Both $Ca_8Mg(SiO_4)_4Cl_2$: Eu^{2+} and Ca₈Mg(SiO₄)₄Cl₂:Ce³⁺, Eu²⁺ emit efficient green and bluish-green lights under near ultraviolet and blue-emitting LED excitations. Hence, they are suitable for the development of fluorescent display devices, efficient green light sources and anticounterfeiting materials.

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