



# High efficient organic ultraviolet photovoltaic devices based on gallium complex

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## ABSTRACT

High efficient organic ultraviolet (UV) photovoltaic devices comprising 4,4',4''-tri-(2-methylphenyl phenylamino) triphenylamine (m-MTDATA) and tris-(8-hydroxyquinoline) gallium (GaQ<sub>3</sub>) as the electron donor and acceptor, respectively, are demonstrated. The m-MTDATA/GaQ<sub>3</sub> bilayer device shows a short-circuit current density of 59.3 μA/cm<sup>2</sup>, a open-circuit voltage of 1.85 V, a fill factor of 0.41, and a power conversion efficiency of 3.74% under illumination of a 1.2 mW/cm<sup>2</sup> 365 nm UV light. And this power conversion efficiency is superior than that of the device based on tris-(8-hydroxyquinoline) aluminum with the same device structure, which is attributed to the high electron mobility of GaQ<sub>3</sub> and the low exciton loss via radiation decay and hence more exciton dissociation in m-MTDATA/GaQ<sub>3</sub> device.

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## 1. Introduction

In the past two decades, there is a growing interest on organic electronic devices, including light-emitting diodes [1], photovoltaic (PV) devices [2], thin film transistors [3], etc., due to their light weight, low cost, and compatibility with flexible substrates compared to inorganic counterparts. Since the demonstration of high efficient exciton dissociation in the donor–acceptor (D–A) heterojunction interface [4], PV devices have drawn particular attention owing to their potential applications in solar cells and photodetectors [5–9]. The PV effect involves the formation of excitons under illumination, the diffusion of excitons to the D–A interface, the dissociation of excitons into electrons and holes, and the collection of electrons and holes at opposite electrodes. To achieve a maximum efficiency, many methods have been adopted to optimize all the four processes involved in the generation of photocarriers [10–12].

The major ultraviolet (UV) radiation reaching the surface of the earth covers wavelength from 300 to 420 nm, and a lot of diseases, such as skin cancer, would be caused by long-term exposure to the UV radiation. Thereby it is necessary to detect the UV radiation intensity for human health. On the other hand, UV PV devices have the potential applications for UV curing monitors, solar astronomy, missile plume detection, and space-to-space transmission. Thus, they are attracting much attention recently. UV PV devices comprising a lot of materials have been reported, such as rear-earth complexes [13–15], Cu(I) complexes [16], and polymer/inorganic

hybrids [17]. Hong et al. [18] reported that PV devices with 4,4',4''-tri-(2-methylphenyl phenylamino) triphenylamine (m-MTDATA) and tris-(8-hydroxyquinoline) aluminum (AlQ<sub>3</sub>) acted as the electron donor and acceptor, respectively. Tris-(8-hydroxyquinoline) gallium (GaQ<sub>3</sub>) has analogous molecular structure and photophysical properties to that of AlQ<sub>3</sub>. The photoluminescent (PL) quantum efficiency of GaQ<sub>3</sub> is demonstrated to be a quarter of that of AlQ<sub>3</sub> [19,20]. Low PL efficiency of the donor and the acceptor is usually considered as one criterion to obtain high power conversion efficiency ( $\eta_p$ ) in PV devices due to the competitive processes of the exciton radiation and exciton dissociation. Moreover, the electron mobility of GaQ<sub>3</sub> is higher than that of AlQ<sub>3</sub> [19,21]. In view of these, GaQ<sub>3</sub> may offer superior performance compared to AlQ<sub>3</sub> in the PV devices. In this paper, UV PV devices with m-MTDATA as the electron donor and GaQ<sub>3</sub> as the electron acceptor, respectively, were fabricated. The m-MTDATA/GaQ<sub>3</sub> bilayer device shows a short-circuit current ( $J_{sc}$ ) of 59.3 μA/cm<sup>2</sup> and a  $\eta_p$  of 3.74%. And the performance is better than that of AlQ<sub>3</sub> with the same device structure, which is attributed to the high electron mobility of GaQ<sub>3</sub> and low exciton loss due to radiation and hence more exciton dissociation in the m-MTDATA/GaQ<sub>3</sub> interface.

## 2. Experimental

Devices were fabricated on precleaned indium tin oxide (ITO)-coated glass substrates with a sheet resistance of 25 Ω/sq. A 50 nm 3,4-polyethylenedioxythiophene:polystyrenesulfonate (PEDOT:PSS) was spin-coated on ITO, followed by drying at 120 °C for

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60 min in vacuum. The PEDOT:PSS layer serves to planarize the ITO, thereby preventing shorts through the thin donor and acceptor layers [22]. Then, m-MTDATA and Gaq<sub>3</sub>, which acted as the electron donor and acceptor, respectively, were thermal evaporated in vacuum chamber at  $3 \times 10^{-4}$  Pa, followed by a 0.5 nm LiF and 100 nm Al cathode. The molecular structures of m-MTDATA and Gaq<sub>3</sub> are shown in Fig. 1. Two devices were fabricated with configurations as follows:

Device A: ITO/PEDOT:PSS/m-MTDATA (30 nm)/Gaq<sub>3</sub> (30 nm)/LiF/Al

Device B: ITO/PEDOT:PSS/m-MTDATA (30 nm)/Alq<sub>3</sub> (30 nm)/LiF/Al

Deposition rates and thickness of the layers were monitored using oscillating quartz monitors. The evaporating rates were kept at 0.5–1 Å/s for organic layers and LiF layer and 10 Å/s for Al cathode, respectively. Absorption spectra of all organic films on quartz substrates were measured with a Shimadzu UV-3101 PC spectrophotometer. PL spectra were measured with a Hitachi F-4500 spectrophotometer. Photocurrent response curve was obtained under irradiance of a 40 μW/cm<sup>2</sup> Xe lamp. Current density–voltage (*I*–*V*) curves were measured by a Keithley 2400 source meter under illumination of 365 nm UV light (1.2 mW/cm<sup>2</sup>) or AM 1.5 solar simulator (100 mW/cm<sup>2</sup>). All the measurements were carried out at room temperature under ambient conditions.

### 3. Results and discussion

Fig. 2 shows the absorption spectra of 30 nm m-MTDATA, 30 nm Gaq<sub>3</sub>, 30 nm Alq<sub>3</sub>, and 30 nm m-MTDATA/30 nm Gaq<sub>3</sub> blend films on quartz substrates. A broad absorption band in the range of 300 to 400 nm in the absorption spectrum was found in the m-MTDATA/Gaq<sub>3</sub> bilayer film, which is a simple superposition of the absorptions of m-MTDATA and Gaq<sub>3</sub>, indicating that the absorptions of m-MTDATA and Gaq<sub>3</sub> are responsible for the photo-induced carrier generation. For reference, the absorption spectrum of 30 nm Alq<sub>3</sub> measured with the same conditions is also presented in Fig. 2. Gaq<sub>3</sub> and Alq<sub>3</sub> have almost the same absorption spectra in the range investigated, although the absorption coefficient of Alq<sub>3</sub> is a little lower than that of Gaq<sub>3</sub>. The photocurrent response curve of Device A is shown in Fig. 3, which reveals a maximum response in the region of 360–370 nm. Thus, a 365 nm UV light was selected as the illumination source for the PV devices.

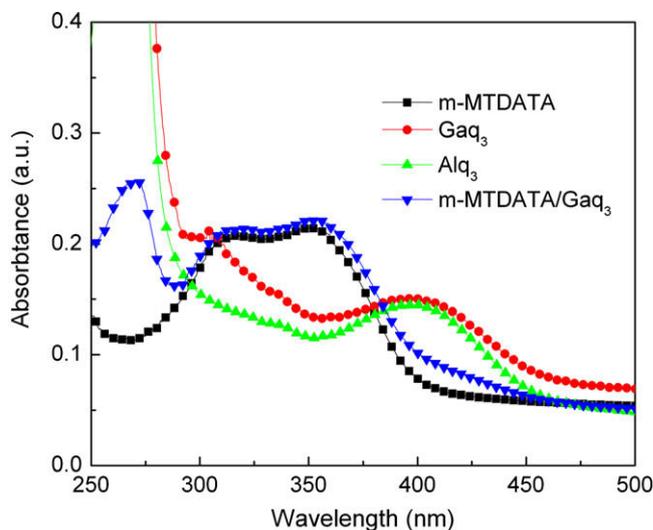
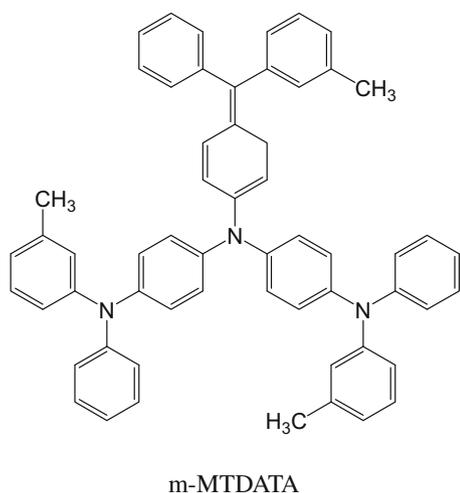


Fig. 2. Absorption spectra of 30 nm m-MTDATA, 30 nm Gaq<sub>3</sub>, 30 nm Alq<sub>3</sub>, and 30 nm m-MTDATA/30 nm Gaq<sub>3</sub> films on quartz substrates.

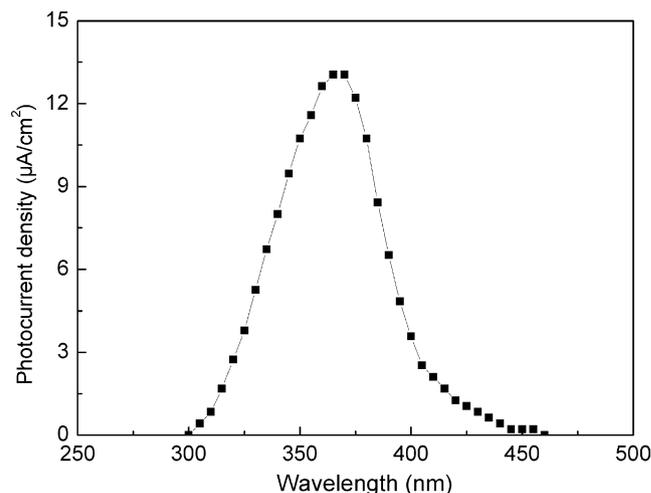


Fig. 3. Photocurrent response of Device A under illumination of a Xe lamp.

Fig. 4 plots the *I*–*V* characteristics of Devices A and B under irradiation of a 365 nm UV light with the intensity of 1.2 mW/cm<sup>2</sup>.

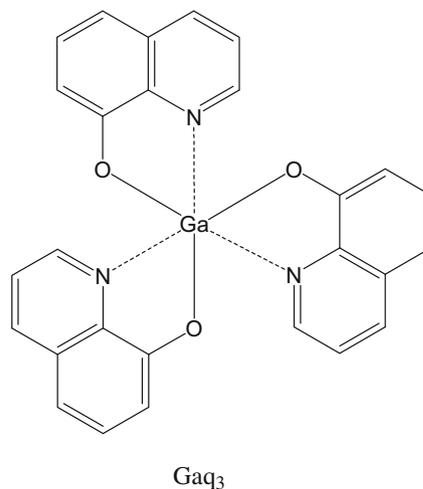


Fig. 1. Molecular structure of m-MTDATA and Gaq<sub>3</sub>.

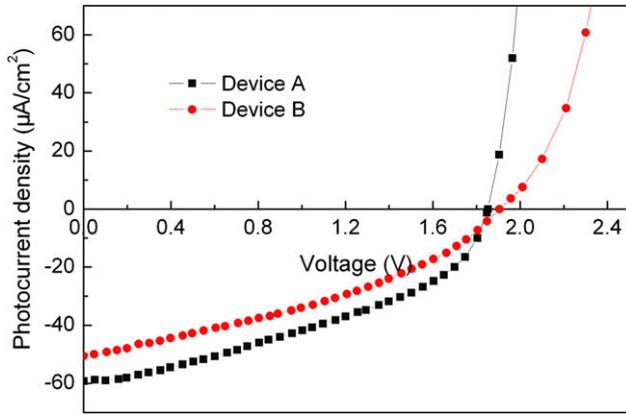


Fig. 4. *I*–*V* characteristics of Devices A and B under illumination of a 1.2 mW/cm<sup>2</sup> UV light.

Among the two devices, Device A presents the superior performance, which shows a  $J_{SC}$  of 59.3  $\mu\text{A}/\text{cm}^2$  and an open-circuit voltage ( $V_{OC}$ ) of 1.85 V. Calculating the fill factor ( $FF$ ) and  $\eta_p$  from the *I*–*V* curve with equations

$$FF = \frac{(J \times V)_{\max}}{V_{oc} \times J_{SC}} \quad (1)$$

$$\eta_p = \frac{(J \times V)_{\max}}{P_{in}} = \frac{FF \times V_{oc} \times J_{SC}}{P_{in}} \quad (2)$$

where  $(J \times V)_{\max}$  is the actual maximum power and  $P_{in}$  is the incident light intensity, a  $FF$  and  $\eta_p$  of 0.41 and 3.74% are achieved, respectively. The  $J_{SC}$ ,  $V_{OC}$ ,  $FF$ , and  $\eta_p$  of Device B are 50.6  $\mu\text{A}/\text{cm}^2$ , 1.91 V, 0.37, and 2.94%, respectively. The performance is much higher than the report by Hong et al. [18], which may be attributed to the improved contact between the electrodes and the organic layers and hence the extraction efficiency of hole and electron from the bulk of the organic films due to the incorporations of the PEDOT:PSS and LiF layers [22,23]. The  $J_{SC}$ ,  $V_{OC}$ ,  $FF$ , and  $\eta_p$  of Device A under 100 mW/cm<sup>2</sup> AM 1.5 solar simulation are 353.1  $\mu\text{A}/\text{cm}^2$ , 1.87 V, 0.28, and 0.18%, while they are 295.5  $\mu\text{A}/\text{cm}^2$ , 1.94 V, 0.27, and 0.16% for Device B, respectively, as shown in Fig. 5. It can be noted that the  $V_{OC}$  of the devices is increased and the  $FF$  is reduced compared to that illuminated by UV light, which are commonly observed in PV devices with increasing irradiation power [22]. The low  $\eta_p$  of Devices A and B is primarily due to they only absorb UV

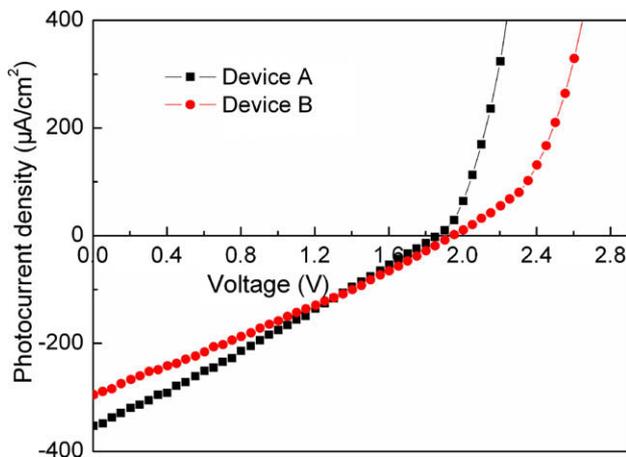


Fig. 5. *I*–*V* characteristics of Devices A and B under 100 mW/cm<sup>2</sup> AM 1.5 illumination.

light (as shown in Fig. 2), which accounts for a small fraction of the solar irradiation. Gaq<sub>3</sub> and Alq<sub>3</sub> have almost the same ionization potential and electron affinity [24,25], thus the difference in energy levels can be ruled out for the better performance of Device A. Meanwhile, a little higher absorption of Gaq<sub>3</sub> as presented in Fig. 2 cannot account for such a distinct distinguished performance between these two devices.

PL quenching of the donor and acceptor is always regarded as an indicator of the performance of PV devices. Fig. 6 depicts the PL spectra of 30 nm Alq<sub>3</sub>, 30 nm Gaq<sub>3</sub>, 60 nm m-MTDATA:Alq<sub>3</sub> (1:1), and 60 nm m-MTDATA:Gaq<sub>3</sub> (1:1) films on quartz substrates with an excitation wavelength of 365 nm. The PL spectra of Alq<sub>3</sub> and Gaq<sub>3</sub> show emission peaks at 512 and 522 nm, respectively. While broadened and red-shifted PL spectra with peaks at about 535 and 537 nm are found in the m-MTDATA:Alq<sub>3</sub> and m-MTDATA:Gaq<sub>3</sub> blend films, respectively, and the PL intensity of the latter is lower than that of the former. The broadened and red-shifted emission of the m-MTDATA:Alq<sub>3</sub> blend film was previously attributed to the hybrid emissions of Alq<sub>3</sub> and the m-MTDATA/Alq<sub>3</sub> interfacial exciplex [18,25]. Thus, emission of the m-MTDATA:Gaq<sub>3</sub> can be similarly ascribed to the hybrid emissions of Gaq<sub>3</sub> and the m-MTDATA/Gaq<sub>3</sub> interfacial exciplex as Gaq<sub>3</sub> has the analogous molecular structure and photophysical properties to that of Alq<sub>3</sub>. No emission of m-MTDATA is found in blend films, indicating that nearly all the m-MTDATA excitons have transferred their electron to the donors Alq<sub>3</sub> and Gaq<sub>3</sub>.

Recently, a two-step model for charge dissociation, proceeding via the formation of interfacial geminate electron–hole pair (or named as interfacial bound radical pair) is proposed [26–28]. According to this model, electron transfer from the donor to the acceptor forms the geminate electron–hole pairs, and the dissociation of the geminate electron–hole pairs generates free charge carriers, while the recombination of the geminate electron–hole pairs results in the radiation of the interfacial exciplex. Thereby the formation of such geminate electron–hole pair rather than energy transfer from the high energy species to the lower energy one in the D–A interface is more critical to the performance of the PV devices [29]. To obtain high  $\eta_p$ , efficient charge transfer in the D–A interface and effective suppression of the geminate electron–hole pair recombination to the ground state should be ensured. The absent emission of m-MTDATA in the blend films (as shown in Fig. 6) suggests that efficient electron transfer from m-MTDATA excitons to Gaq<sub>3</sub> and Alq<sub>3</sub> molecules takes place in the m-MTDATA/Gaq<sub>3</sub> and m-MTDATA/Alq<sub>3</sub> interfaces. The remained emissions of Gaq<sub>3</sub>

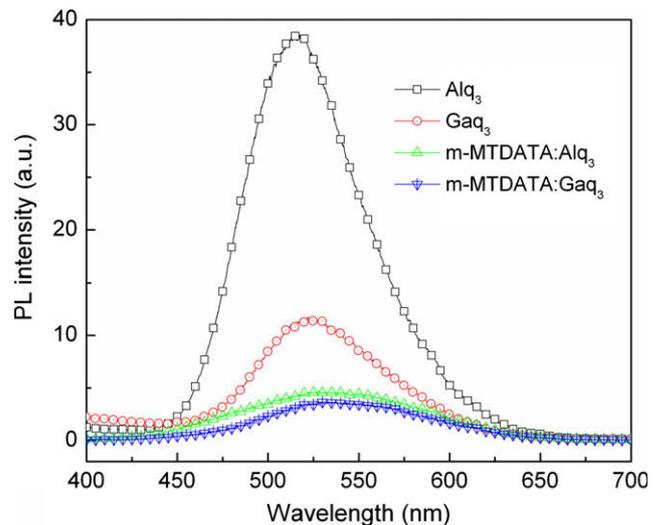


Fig. 6. PL spectra of 30 nm Alq<sub>3</sub>, 30 nm Gaq<sub>3</sub>, 60 nm m-MTDATA:Alq<sub>3</sub> (1:1), and 60 nm m-MTDATA:Gaq<sub>3</sub> (1:1) films on quartz substrates excited at 365 nm.

or Alq<sub>3</sub> in the blend films indicate that some Gaq<sub>3</sub> or Alq<sub>3</sub> excitons are precluded in the formation of geminate electron–hole pair, and the appearance of exciplexes emission illuminates the recombination of the geminate electron–hole pair. The lower PL emission of the m-MTDATA:Gaq<sub>3</sub> blend film than that of m-MTDATA:Alq<sub>3</sub> indicates low radiation loss in Device A. On the other hand, after the dissociation of the geminate electron–hole pairs, higher electron mobility of Gaq<sub>3</sub> favors the extraction of electron and consequence the dissociation of the geminate electron–hole pairs. Therefore, the combination of low radiation loss and high electron mobility of Gaq<sub>3</sub> is the main cause of the superior performance of Device A than B.

#### 4. Conclusion

In summary, high efficient UV PV devices based on m-MTDATA as the donor and Gaq<sub>3</sub> as the acceptor, respectively, were demonstrated. The m-MTDATA/Gaq<sub>3</sub> bilayer device shows a  $J_{SC}$  of 59.3  $\mu\text{A}/\text{cm}^2$ , a  $V_{OC}$  of 1.85 V, and a  $\eta_p$  of 3.74% under illumination of a 1.2  $\text{mW}/\text{cm}^2$  at 365 nm. These findings indicate that Gaq<sub>3</sub> has the potential application in high efficient UV photodetector [30]. Moreover, the performance is better than that of the device based on Alq<sub>3</sub> with the same structure, which is predominantly attributed to the high electron mobility of Gaq<sub>3</sub> and the low radiation loss in Gaq<sub>3</sub> device.

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