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Comparative Study of Properties of ZnO/GaN/Al₂O₃ and ZnO/Al₂O₃ Films Grown by Low-Pressure Metal Organic Chemical Vapour Deposition *

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ZnO films were deposited by low-pressure metal organic chemical vapour deposition on epi-GaN/Al₂O₃ films and c-Al₂O₃ substrates. The structure and optical properties of the ZnO/GaN/Al₂O₃ and ZnO/Al₂O₃ films have been investigated to determine the differences between the two substrates. ZnO films on GaN/Al₂O₃ show very strong emission features associated with exciton transitions, just as ZnO films on Al₂O₃, while the crystalline structural qualities for ZnO films on GaN/Al₂O₃ are much better than those for ZnO films directly grown on Al₂O₃ substrates. Zn and O elements in the deposited ZnO/GaN/Al₂O₃ and ZnO/Al₂O₃ films are investigated and compared by x-ray photoelectron spectroscopy. According to the statistical results, the Zn/O ratio changes from Zn-rich for ZnO/Al₂O₃ films to O-rich for ZnO/GaN/Al₂O₃ films.

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There is an increasing demand for photonic devices in blue and ultraviolet ranges. $^{[1,2]}$ ZnO, with a wide direct band-gap of $3.37\,\mathrm{eV}$ at room temperature, is an attractive material for many applications, such as opto-electronics devices, $^{[3]}$ surface acoustic wave $^{[4]}$ and transparent electrode, $^{[5,6]}$, etc. In particular, the ultraviolet lasing has been observed in ZnO thin films at room temperature, $^{[7-10]}$ which attracts much attention in this area.

ZnO-based heteroepitaxial structures are usually grown on sapphire substrates. Although ZnO films grown on Al_2O_3 have shown promising optical properties, the crystal quality of ZnO layers still needs improvement due to the presence of high density of defects nearly at the interface^[11] as a consequence of a large lattice mismatch of 18% in the ZnO/Al $_2O_3$ heterostructure.

Obviously, a close matching between the ZnO and GaN crystal lattices can be used for obtaining high-quality heteroepitaxial structures of ZnO on GaN, so epitaxial GaN (epi-GaN) grown Al₂O₃ is now available as a substrate to grow ZnO films.

In this Letter, we report the examination of relationship between the crystalline qualities and optical features determined by comparing the properties of ZnO layers on epi-GaN/Al $_2$ O $_3$ and Al $_2$ O $_3$ by low-pressure metal organic chemical vapour deposition

(LP-MOCVD). The film properties were determined by x-ray diffraction (XRD), atomic force microscopy (AFM), x-ray photoelectron spectroscopy (XPS), and photoluminescence (PL) spectroscopy.

In the experiment, the epitaxial layers of (0001) GaN on (0001)Al₂O₃ were obtained by metal organic chemical vapour deposition (MOCVD) using a lowtemperature buffer layer technique.^[12] The GaN layer thickness on sapphire was about $3 \mu m$. These structures were transferred into rotating disc vertical reactor of LP-MOCVD without preliminary cleaning and polishing procedures. The source materials were Diethylzinc (DEZn) and O₂, both with high purity of 99.999%, the carrier gases were argon and nitrogen. The growth condition included 1.50 sccm of argon (saturated with DEZn vapour) flow, 5sccm of oxygen flow, 10 sccm of nitrogen flow, 0.65 Pa of reactor gas pressure. After heating at 700°C for 10 min, ZnO films were then deposited at the substrate temperature of 600°C for 40 min. The as-grown film was further annealed in oxygen plasma atmosphere at 700°C for 10 min. The final ZnO layer thickness was about $3 \,\mu\mathrm{m}$. For comparative analysis, the ZnO film was grown directly on the Al₂O₃ substrate under the same growth condition.

We used a SIEMENS D8/C2 x-ray diffractometer and a Rigaku DMAX 2400 x-ray diffractometer to

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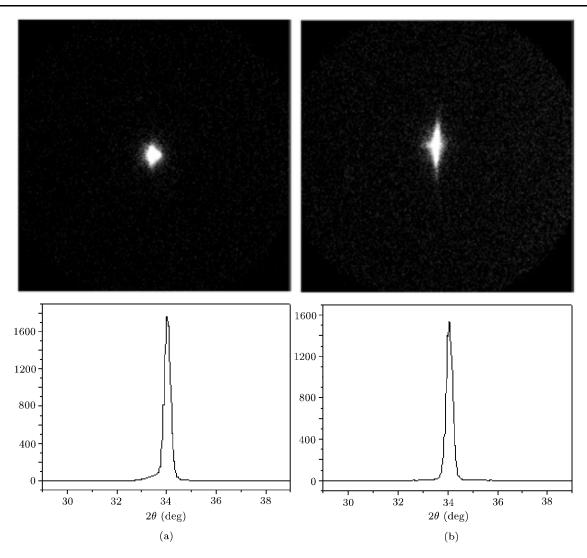


Fig. 1. XRD spectra and spots of (a) $\rm ZnO/GaN/Al_2O_3$ film and (b) $\rm ZnO/Al_2O_3$ film.

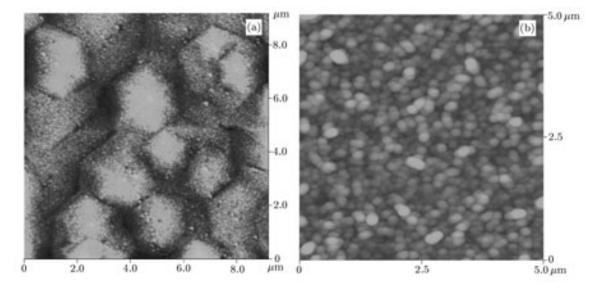


Fig. 2. Surface AFM image of (a) $\rm ZnO/GaN/Al_2O_3$ film and (b) $\rm ZnO/Al_2O_3$ film.

investigate the crystal quality and integrality. Surface morphology was investigated by Digital Nanoscope IIIa AFM with normal silicon nitride tip in the contact mode. The PL spectrum was measured by a 325-nm He-Cd laser. The PL signal from the sample was filtered by a monochromator and picked up by a CCD detector. The power arriving at the sample was about 3 mW with a beam diameter of $200\,\mu\mathrm{m}$. For the low-temperature measurement, the sample was mounted in a closed-cycle refrigerator.

x-ray diffraction measurement of θ -2 θ scan by the SIEMENS D8/C2 diffractometer was performed on ZnO films as shown in Fig. 1. Dominant diffraction spots at $2\theta = 34.6^{\circ}$ due to the ZnO (0002) peak could be observed respectively. It is obvious that the spot size of the ZnO/GaN/Al₂O₃ film was smaller than that of the ZnO/Al₂O₃ film, which indicates much better crystal quality for the ZnO/GaN/Al₂O₃ film. The asymmetry of spot for the ZnO/GaN/Al₂O₃ film implies that the ZnO spot on the right-hand side and GaN spot on the left-hand side were overlapping due to the close crystal structure. The XRD $2\theta - \omega$ scan was also performed by the DMAX 2400 x-ray diffractometer. The rocking curve of the (0002) reflection showed that the FWHM of the peak was 0.39° for the ZnO/GaN/Al₂O₃ film and 0.6° for the ZnO/Al₂O₃ film, which indicates that the mosaicity of the ZnO/GaN/Al₂O₃ film was really smaller and the crystal quality was better.

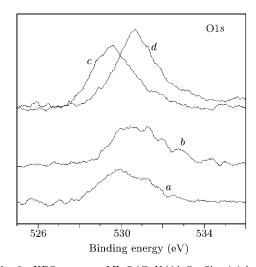


Fig. 3. XPS spectra of $\rm ZnO/GaN/Al_2O_3$ film (a) before and (c) after the, etching, and $\rm ZnO/Al_2O_3$ film (b) before and (d) after the, etching.

The AFM measurements were performed to study the differences between the two epitaxial ZnO films in the surface morphology. Figure 2(a) shows the AFM images for the ZnO/GaN/Al₂O₃ film scanned over an area of $10 \times 10 \,\mu\text{m}$. We can find many regular hexagonal grain particles in the AFM image. The average grain diameter of the film was about $1 \,\mu\text{m}$. Figure 2(b) shows the image of the ZnO/Al₂O₃ film scanned

over an area of $5\times 5~\mu m$. The mean island shape grain particles with 50-nm sizes were found on the surface. It can be concluded that the regular hexagonal column growth of the ZnO film strictly followed the hexagonal structure of the GaN epi-layer, which was more excellent than the island growth of ZnO film on the Al_2O_3 substrate.

x-ray photoelectron spectra (XPS) were performed to investigate the elements and stoichiometry of the ZnO film grown on the epi-GaN layer and Al_2O_3 . In order to avoid the influence of surface absorption in the atmosphere, Ar ion, etching was performed for about 30 min with an, etching rate of $0.5 \,\mathrm{nm}~\mathrm{min}^{-1}$. Curves a and c in Fig. 3 represent the O 1s spectra of the ZnO/GaN/Al₂O₃ films before and after, etching: curves b and d in Figs. 3 for ZnO/Al_2O_3 films before and after, etching. We know that the typical O 1speak in the ZnO film can be considered as the combination of three nearly Gaussian peaks, centred at 530, 531.25, 532.4 eV, respectively. [13] The O 1 s spectrum at 530 eV is attributed to O²⁻ ions on the wurtzite structure of hexagonal Zn^{2+} ion array, the O 1 s spectrum at $531.25 \,\mathrm{eV}$ is associated with O^{2-} ions in the oxygen deficient regions, which is related to the concentration of oxygen vacancies, and the high binding energy component located at 532.4 eV is usually attributed to the presence of loosely bound oxygen on the surface of the ZnO film. We could find from Fig. 3 that O 1s shifted toward the low-energy direction for the $ZnO/GaN/Al_2O_3$ film and the peak at 531.25 eV was difficult to be observed. This implies that the Zn and O elements in the ZnO film existed mostly with the Zn-O bond. From the statistical results XPS, we could obtain that the atomic ratio of Zn/O was nearly 1:1.03 for the $ZnO/GaN/Al_2O_3$ film and 1:0.95 for the ZnO/Al_2O_3 film.

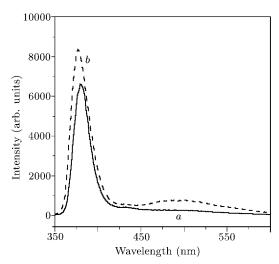


Fig. 4. PL spectrum of (a) $ZnO/GaN/Al_2O_3$ film and (b) ZnO/Al_2O_3 film.

Optical properties of ZnO films were characterized at room temperature by PL spectroscopy in Fig. 4. Strong ultraviolet (UV) emission coming from exciton emission could be observed in both the samples. However, the deep-level emission at 520 nm was observed in the ZnO/Al₂O₃ film. In general, this deep green emission is believed to come from oxygen vacancies, interstitial zinc or zinc vacancies. [14–16] From the spectrum of the ZnO/GaN/Al₂O₃ film, the absence of the deep-level emission peak indicates that the ZnO film grown on epi-GaN has better optical quality and fewer interior defects.

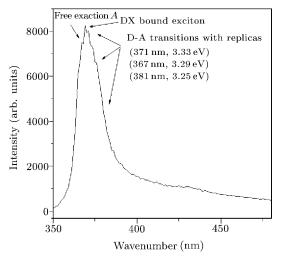


Fig. 5. Low-temperature PL spectrum of ${\rm ZnO/GaN/Al_2O_3}$ film at 20 K.

For further study of the optical properties, the PL measurement at $20\,\mathrm{K}$ was performed on the $\mathrm{ZnO/GaN/Al_2\,O_3}$ film. As shown in Fig. 5, the peak energy positions at 3.376 and $3.361\,\mathrm{eV}$ correspond to the A-band free exciton emission and the D°X bound exciton transition. The peak at $3.33\,\mathrm{eV}$ may pertain to the donor-acceptor pair transition while the peaks at 3.29 and $3.25\,\mathrm{eV}$ might be ascribed to LO phonon

replicas. The observation of the A-exciton transition and its high intensity showed the better quality of the $\rm ZnO/GaN/Al_2O_3$ film.

In summary, we have fabricated high-quality epitaxial $\rm ZnO/GaN$ heterostructures on sapphire. Comparative studies of XRD, AFM, XPS and PL spectra of $\rm ZnO/Al_2O_3$ films implies that the $\rm ZnO/GaN/Al_2O_3$ film shows substantial improvement in crystal quality, growth mode, defect density and luminescence properties. In addition, the matching of thermal and optical properties and the compatibility of these heterostructures provide new opportunities for the fabrication of hybrid $\rm ZnO/GaN$ optoelectronic devices on sapphire.

References

- [1] Zhou J and Zhang G Y 2002 Chin. Phys. Lett. 19 599
- 2] Pan Z, Li L H, Xu Y Q, Zhang W, Lin Y W, Zhang R K, Zhong Y and Ren X M 2001 Chin. Phys. Lett. 18 1249
- [3] Jeong S, Kim B and Lee B 2003 Appl. Phys. Lett. 16 2625
- [4] Kazunori M, Yasushi K, Yukinobu K, Koji Y, Masanobu K and Azuma S 1997 Jpn. J. Appl. Phys. 36 1453
- [5] Carlotti G and Socino G 1987 Appl. Phys. Lett. 51 1889
- [6] Koike J, Shimoe K and Ieki H 1993 Jpn. J. Appl. Phys. 32 2337
- [7] Tang Z, Wong G, Yu P, Kawasaki M, Ohtomo A, Koinuma H and Segawa Y 1998 Appl. Phys. Lett. 72 3270
- [8] Chen Y, Bagnall D, Zhu Z, Sekiuchi T, Park K, Hiraga K, Yao T, Koyama S, Shen M and Goto T 1997 J. Cryst. Growth 181 165
- [9] Bagnall D, Chen Y, Zhu Z, Yao T, Koyama S, Shen M and Goto T 1997 Appl. Phys. Lett. 70 2230
- [10] Segawa Y, Ohtomo A and Kawasaki M 1997 Phys. Status Solidi B 202 669
- [11] Chen Y F, Bagnall D M and Ko H J 1998 J. Appl. Phys. 84 3912
- [12] Lundin W V, Pushnyi B V and Usikov A S 1997 Inst. Phys. Conf. Ser. 155 319
- [13] chen M, Wang X and Yu Y H 2000 Appl. Surf. Sci. 158
- [14] Wang X Q, Yang S R, Wang J Z, Li M T, Jiang X Y, Du G T, Liu X and Chang R 2001 J. Cryst. Growth 226 123
- [15] Wang J Z et al 2002 Chin. Phys. Lett. 19 581
- [16] Vanheusden K, Warren W L and Seager C H 1996 J. Appl. Phys. 79 7983